IV.A.1g Fundamental Studies of Advanced High-Capacity, Reversible Metal Hydrides

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Objectives

The objective of this project is to develop a new class of reversible materials that have the potential to meet the DOE 2010 kinetic and system gravimetric storage capacity targets. Current investigations include:

- The investigation of the hydrogen storage properties of magnesium nano-confined in carbon aerogels.
- The study of novel, high hydrogen capacity, borohydrides that can be reversibly dehydrogenated at low temperatures.
- The development of a method for the hydrogenation of Al to alane, AlH₃, and/or LiH/Al to LiAlH₄ at moderate pressures in non-conventional solvents.

Technical Barriers

This project addresses the following technical barriers from the Hydrogen Storage section of the Hydrogen, Fuel Cells and Infrastructure Technologies Program Multi-Year Research, Development and Demonstration Plan:

(A) System Weight and Volume

- (E) Charging/Discharging Rates
- (J) Thermal management
- (P) Lack of Understanding of Hydrogen Physisorption and Chemisorption

Technical Targets

The work on this project is currently exclusively devoted to materials that have demonstrated available hydrogen capacities of 7-17 wt% hydrogen. We have developed a system of for the full reversible dehydrogenation of $Mg(BH_4)_2$ to MgB_2 that has shown a record, >12 wt% reversible hydrogen capacity but requires further development to meet kinetic performance targets within the target temperature. We have also developed a highly efficient method of the direct hydrogen capacity, rapid dehydrogenation at 120-150°C) at moderate pressure and room temperature that has been estimated to have a well-to-tank (WTT) energy efficiency that approaches 60% and thus meets U.S. DOE targets.

Accomplishments

Nano-Confined Mg in Carbon Aerogels

- Confirmed MgH₂ incorporation in aerogel by transmission electron microscopy (TEM) imaging, energy dispersive X-ray spectroscopy, and X-ray diffraction (XRD).
- Obtained 23 wt% MgH₂ loadings in carbon aerogels with larger pore sizes.
- Determined the rate of dehydrogenation at 252°C is >5 times faster than the initial rate found for ball milled MgH₂ and that it remains approximately the same over four cycles of dehydrogenation/rehydrogenation.
- Established that thermodynamics of the dehydrogenation of MgH₂ are not altered by nano-confinement.

Novel Borohydrides

- Achieved full, ~14 wt% re-hydrogenation of MgB₂ to Mg(BH₄)₂ through a high pressure method that avoids formation of highly stable, MgB₁₂H₁₂
- Confirmed that the only major product of high pressure hydrogenation is Mg(BH₄)₂ by magic angle spinning (MAS) ¹¹B nuclear magnetic resonance (NMR) spectroscopy.

 Filed patent application for method for hydrogenation of MgB₂ to Mg(BH₄)₂.

Hydrogenation in Non-Conventional Solvents

- Developed a method for the room temperature, direct hydrogenation of Ti-doped LiH/Al in liquefied dimethyl ether under 100 bar of H₂.
- Optimized synthesis process such that charged, Tidoped LiAlH₄ is obtained in >95% yield.
- Determined the WTT energy efficiency our direct synthesis process approaches the 60% U.S. DOE target.
- Filed patent application for our direct synthesis process.

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Introduction

The development of high capacity, hydrogen storage materials that can be recharged under moderate conditions is a key barrier to the realization of a hydrogen economy. Towards this goal, we are exploring new approaches for homogeneous incorporation of high hydrogen capacity metal hydrides into carbon aerogels. Intercalation into these nano-porous materials may change the thermodynamics and/or enhance the kinetics of the reversible hydrogenation thus bringing previously non-viable, metal hydrides such as MgH₂ into the realm of practical hydrogen storage materials.

We are also developing high hydrogen capacity borohydrides as hydrogen storage materials. Our initial focus was on anionic transition metal complexes. The anionic character of these compounds was found to result in an increased stability and a reduced volatility when compared to neutral transition metal borohydride complexes. However, in the course of our studies, we found a method whereby the fully reversible dehydrogenation Mg(BH₄)₂ to MgB₂ can be achieved. In consideration of the >14 wt% hydrogen that is potentially cyclable with this system, it has become the focus of our efforts on borohydrides.

The dehydrogenation of AlH₃ to Al gives 10 wt% H₂. With additives, controllable dehydrogenation is possible at acceptable rates below 100°C. The low (<10 kJ/mol H₂) Δ H_{dehyd} mandates very high pressures for charging at ambient or higher temperatures. Supercritical fluids (SCFs) have different physical properties than gases. It has been found that requisite pressures hydrogenations are reduced in SCFs (i.e. hydrogenations of organic molecules and NaH/Al to NaAlH₄). In collaboration with the University of New Brunswick, we are developing new approaches utilizing supercritical fluids recharging AlH₂. We are also exploring the direct synthesis of LiAlH_4 in liquefied dimethyl ether, Me₂O. LiAlH_4 releases hydrogen according to the two reactions seen in equations 4 and 5. Although the second reaction is endothermic ($\Delta \text{H} = +25 \text{ kJ/mol H}_2$), the first reaction

$$3 \operatorname{LiAlH}_{4} \rightarrow \operatorname{Li}_{3} \operatorname{AlH}_{6} + 2\operatorname{Al} + 3\operatorname{H}_{2}$$

$$\tag{4}$$

$$\text{Li}_{3}\text{AIH}_{6} \rightarrow 3 \text{ LiH} + \text{AI} + 1.5\text{H}_{2} \tag{5}$$

is exothermic ($\Delta H = -10 \text{ kJ/mol H}_2$) [1]. Since entropic change is strongly positive for hydrogen release, the first reaction is thermodynamically irreversible under all practical conditions and it has been widely accepted that LiAlH, cannot be recharged and there have been no reports of its use for reversible hydrogen storage. However, the unfavorable thermodynamics of Eq. 4 may be altered by carrying out the reaction in solution, with the solvation of LiAlH₄ contributing to an (ideally) endothermic ΔH value. As early as 1963, Ashby *et al.* reported that a mixture of LiH and activated Al in THF or diglyme solvent reacted with 350 bar H₂ at 120°C to produce LiAlH₄ [2]. An improvement of the synthesis of the tetrahydrofuran (THF) adduct has recently achieved by researchers at Brookhaven National Laboratory by using Ti additives to significantly lower the required pressure and temperature [3]. A variation of this synthesis was previously reported by Ritter et al. in a procedure that involves the high-energy ball milling of a LiH/Al mixture under H₂ pressure in the presence of THF [4]. While the methods of both Ashby and Ritter result in the ultimate formation of LiAlH, from LiH, Al and H₂, they are impractical because of the requirements of high temperature, high pressure, and/or mechanical energy during the synthesis as well as subsequent heating removal the THF solvent (vacuum drying at 60°C for several hours). Ashby noted that it is very difficult to remove the final vestiges of THF. The prolonged baking in vacuo required to remove THF will necessarily compromise the integrity of a Ti-activated product. Clearly, the key to practically is the identification of a solvent that is sufficiently coordinating to support the nascent LiAlH, in the reaction environment, yet which is also volatile enough to be removed easily once it has served this purpose.

Approach

We have developed a novel approach for the intercalation of Mg in carbon aerogel materials. This procedure achieves high, cyclable magnesium loadings in carbon aerogels without host degradation. Since this method is performed at relatively low temperatures, the pore size of the aerogel material is preserved without degradation. The high loadings are allowing us to conduct highly reliable studies of the effects of nano-confinement on the kinetics and thermodynamics of the dehydrogenation of MgH₂.

We have found that group I and II salts of anionic transition metal borohydride complexes have several improvements over neutral transition metal borohydrides: including higher (9-13 wt%) hydrogen content; reduced volatility; increase stability; and for some, complexes such as $Na_2Zr(BH_4)_6$, very low levels of diborane evolution during the dehydrogenation. Our studies are currently focused on determining whether the thermodynamic parameters of these complexes will allow them to undergo reversible dehydrogenation. In the course of these investigations we have found that that ball-milled mixtures of MgB₂ and catalytic additives undergo full hydrogenation to $Mg(BH_4)_2$ at high pressures. We are currently conducting detailed studies of the reversible dehydrogenation >12 wt% H₂ from $Mg(BH_{4})_{2}$.

Our previous efforts have resulted in the development a SCF medium for the hydrogenation of Al that that facilitates the hydrogenation of Al to AlH_3 in ~5% yield. Attempts to achieve higher yields by conducting the hydrogenation in alternative SCFs and utilizing variety initiator and catalysts are currently underway.

We are also exploring the synthesis of LiAlH_4 in liquefied Me₂O. Dimethyl ether should perform well as the solvent for this reaction since it is more polar and volatile than diethyl ether, it forms a strong complex with Li⁺, and it evaporates quickly at room temperature (boiling point = -24°C).

Results

Task 1. Characterization of the Active Titanium Species in Ti Doped NaAlH₄

This task has been completed.

Task 2. Spectroscopic Studies of Complex Hydrides

This task has been completed.

Task 3. Thermodynamic Properties of Complex Hydrides

This task has been completed.

Task 4. Kinetic Enhancement of "Thermodynamically Tuned" Binary Hydrides

Previously we reported high (9-16 wt%) MgH_2 loadings of carbon aerogel without host degradation are obtained using organometallic method. TEM imaging studies of our magnesium nano-confined aerogel have been completed in collaboration with the California Institute of Technology. These studies point to the nanosize of the Mg and MgH₂ particles. We also examined the effect of pore size of carbon scaffold materials on the MgH₂ loading. Results showed that higher MgH₂ loadings can be obtained with materials that have larger pore sizes (Table 1).

TABLE 1.	Effect of	Increasing	Pore Si	ize on	MgH,	Loading

Scaffold	Activated Carbon	13 nm Carbon Aerogel	25 nm Carbon Aerogel
Pore Size (nm)	<2	13	25
Pore Volume (cm ³ /g)	0.58	0.80	1.38
MgH ₂ Loading (wt%)	4.2	17	23

We have found that the rate of dehydrogenation of the nano-confined MgH_a at 252°C is >5 times faster than the initial rate found for ball milled MgH₂ and comparable to those found for nano-confined MgH_a in carbon aerogels that was prepared by alternative methods at HRL (Figure 1). We have also conducted comparative cycling studies of MgH₂ incorporated into aerogel granules and aerogel powder. Dehydrogenation was carried out at 325°C under vacuum and hydrogenation was performed under 9-13.5 MPa at 177-325°C for 6-20 hours. We found that with aerogel granules, the dehydrogenation rates remained approximately the same over four cycles and that they contained about 12 wt% reversible MgH₂ (Figures 2 and 3). Cycling tests of aerogel powder showed an obvious degradation of the dehydrogenation kinetics, and aggregation of Mg was observed.

To improve the kinetics of dehydrogenation and re-hydrogenation of the nano-confined MgH_2 we incorporated Ti-catalyst in the carbon aerogel. The Ti(0) catalyst was synthesized in situ through the reaction of TiCl₄·2THF with LiAlH₄. We developed two procedures to carry out this synthesis. The first involved initial preparation of Ti(0) incorporated aerogel that was next loaded with MgBu₂ and finally hydrogenated to MgH₂. The second procedure involved the reciprocal impregnation of Ti(0) into MgH₂. We found that the



FIGURE 1. The rate of dehydrogenation at 252° C is >5 times faster than the initial rate found for ball milled MgH_a.



FIGURE 2. Reversible dehydrogenation kinetics of Mg intercalated carbon aerogels through four cycles.

initially prepared materials containing Ti catalyst did not show any kinetic improvement. In fact, the amount of H_2 evolution actually decreased when compared to the material without the Ti species.

We have started measurements of pressurecomposition isotherms to accurately determine if the thermodynamic properties of MgH_2 are changed upon nano-confinement. The equilibrium pressure of MgH_2 nano-confined in 13 nm aerogel at 250°C agrees with database value of 0.9 atm for bulk MgH_2 . This preliminary result indicates that the enthalpy of dehydrogenation of the nano-confined MgH_2 is 70 KJ/mol which is not significantly different than that of bulk MgH_2 .

Task 5. Synthesis and Evaluation of Novel Borohydrides

We have found that the alkali metal salts of anionic transition metal borohydride complexes such $LiMn(BH_4)_4$ undergo rapid dehydrogenation at moderate (<150°C) temperatures while evolving high (>3) wt% of hydrogen. In some cases, only very minor the amounts of the undesirable, diborane side product are produced. However, we have not found the dehydrogenation of any of these materials to be reversible. In order to see if the re-hydrogenation can be achieved even at high pressures, we conducted experiments in the range of 900 bar in collaboration with Sandia National Laboratories.

Experiments involving the attempted hydrogenation of various ball-milled mixtures alkali metal borides or hydrides; transition metal hydrides and borides; and catalytic additives proved unsuccessful. However, hydrogen uptake was found to occur in experiments with mixtures of MgB₂; transition metal hydrides; and catalytic additive. Analysis of the product mixture by XRD showed that the transition metal component was unreactive while the MgB₂ was hydrogenated to Mg(BH₄)₂. The same result was achieved when starting with ball-milled mixtures of MgB₂ and the catalytic additive at 400°C and 900 atm. A previous effort



FIGURE 3. The MgH, intercalated carbon aerogel contains about 12 wt% of reversible MgH,.

within the Metal Hydride Center of Excellence headed by researchers at General Electric showed $Mg(BH_4)_2$ undergoes stepwise dehydrogenation to give MgB_2 while releasing 14.8 wt% H_2 as seen in equations 1-3. Thermodynamic considerations indicate that it

$$6 \text{ Mg(BH}_{4})_{2} \rightarrow 5 \text{ MgH}_{2} + \text{Mg(B}_{12}\text{H}_{12}) + 13 \text{ H}_{2}\uparrow$$
(1)

$$5 \text{ MgH}_2 + \text{Mg(B}_{12}\text{H}_{12}) \rightarrow 5 \text{ Mg} + 5 \text{H}_2 \uparrow + \text{Mg(B}_{12}\text{H}_{12})$$
 (2)

$$5 \text{ Mg} + \text{Mg}(\text{B}_{12}\text{H}_{12}) \rightarrow 6 \text{ MgB}_2 + 6 \text{ H}_2 \uparrow$$
(3)

should be possible to hydrogenate MgB_2 to $Mg(BH_4)_2$ at moderate temperatures and pressures. However, experimentally it has proven possible to reverse only the processes in equations 2 and 3 at temperatures lower than 500°C. It has been suggested that this is due to the high kinetic stabilization of $MgB_{12}H_{12}$. Thus our finding of full hydrogenation to $Mg(BH_4)_2$ is a breakthrough in the effort to develop this compound as a hydrogen storage material. We have also confirmed its identity by MAS¹¹B NMR spectroscopy. This analysis is very important as previous studies of the hydrogenation of MgB have sound that the obtained $MgB_{12}H_{12}$ is amorphous and thus cannot be detected by XRD. By contrast, ¹¹B NMR is able to detect all boron-containing materials in the product mixture. As seen in Figure 4, only one major boroncontaining product is observed. The chemical shift of -41 ppm allows its clear identification as $Mg(BH_4)_2$ as the shift exactly matches that which we have observed for an authentic sample of $Mg(BH_4)_2$. Only a very minor signal is observed for $MgB_{12}H_{12}$, at -24 ppm indicating that it represents <5% of the product mixture. It is not clear at this point if the high-pressure low-pressure reaction proceed by the same pathway. Thus the minor amount of MgB₁₂H₁₂ that was obtained could be either a small amount of an unreacted intermediate in the catalyzed dehydrogenation pathway or, alternatively, it could be a



FIGURE 4. MAS ^{11}B NMR MgB_{2} + 2 mol% additive hydrogenated under 950 bar at 400 °C.

small amount of product resulting from the competing, standard uncatalyzed pathway.

As seen in Figure 5, thermal desorption of $Mg(BH_4)_2$ synthesized from high pressure hydrogenation of MgB_2 yields 11.5 wt% hydrogen. If the catalyst contribution to sample weight is not taken into consideration, 12.7 wt% hydrogen is realized, representing about 86% conversion of MgB_2 to $Mg(BH_4)_2$. XRD analysis of the synthesized product shows the presence of MgO, which contributes to the low weight percent hydrogen.

Task 6. Recharging of Light Metal Hydrides in Non-Conventional Media

A variety of alternative SCFs, initiators and catalysts were screened in attempts to improve the yield of AlH_3 from the hydrogenation of Al in SCFs. However, we have been unable to improve upon the ~5% (surface only?) yield we reported last year.

In a more promising development, we have found that LiAlH₄ can be synthesized in liquefied dimethyl ether (Me₂O) at room temperature from the hydrogenation of a mixture of LiH and Al in the presence of very low amounts of Ti catalyst (0.5-0.2 mol%). The method requires only moderate pressures of H₂ (80-100 bar). On completion of reaction, the volatile Me₂O solvent and excess H₂ are vented leaving only fine, dry Ti-doped LiAlH, that, as seen in Figure 6, can be clearly identified by XRD. We have found that improved yields are obtained upon lowering the Ti doping level. As seen in Figure 7, the amount of hydrogen obtained from the materials synthesized by our methods upon heating between 80 and 150°C dramatically increases as the doping level is decreased from 2.0 to 0.2 mol%. At the lower Ti loading a > 7 wt% H₂ is obtained which corresponds to a yield of ca. 95% [5]. Thus we have found a highly efficient method of the direct synthesis of LiAlH₄ at low pressure and room temperature. In further consideration that the energy for compression of Me₂O and H₂ is about 1/5



FIGURE 5. Thermal desorption curve for high pressure, direct synthesized $Mg(BH_4)_2$.



FIGURE 6. XRD patterns of LiAlH₄ samples: (a) International Centre for Diffraction Data[®] ref; (b) ball milled (2.0 mol% TiCl₃); (c) fully dehydrogenated (0.2 mol% TiCl₃); (d) rehydrogenated (2.0 mol% TiCl₃); (e) rehydrogenated (0.2 mol% TiCl₃).

that of H_2 and that $LiAlH_4$ has a high solubility in Me_2O , a preliminary analysis conducted at Argonne National Laboratory has indicated that WTT energy efficiency of our regeneration process is great than 60% and thus in line with U.S. DOE targets.

Conclusions

In summary, using our low temperature, "neat organometallic" method, we have achieved high, 9-16 wt% Mg loadings in carbon aerogels without host degradation. Higher MgH₂ loadings obtained with



FIGURE 7. Temperature-programmed desorption plots for LiAlH₄ samples: (a) as-received; (b) ball milled with 2.0 mol% TiCl₃; (c) rehydrogenated (2.0 mol% TiCl₃); (d) rehydrogenated (1.0 mol% TiCl₃); (e) rehydrogenated (0.5 mol% TiCl₃); and (f) rehydrogenated (0.2 mol% TiCl₃).

materials that have larger pore sizes. In addition, our studies have concluded that nano-confinement of MgH_2 improves the dehydrogenation kinetics, but does not appear to effect its thermodynamic properties, most notably, the enthalpy of dehydrogenation.

We have also achieved the full re-hydrogenation of MgB_2 to $Mg(BH_4)_2$ through a high pressure method that avoids formation of highly stable, $MgB_{12}H_{12}$ that typically precludes a full, 14 wt% hydrogenation. This result was first indicted by XRD and also confirmed by MAS ¹¹B NMR spectroscopy studies which show that the only major product of the hydrogenation is $Mg(BH_4)_2$.

Finally, we have developed a method whereby 95% yields of fully charged, Ti-doped LiAlH₄ are obtained from the direct hydrogenation of Ti-doped LiH/Al. The method involves room temperature hydrogenation in liquefied dimethyl ether under 100 bar of H₂. The WTT energy efficiency of a LiAlH₄-based hydrogen system utilizing liquid dimethyl ether as a re-hydrogenation medium approaches the 60% US DOE target.

Future Directions

Nano-confined Mg in Carbon Aerogels

- Determining dehydrogenation and re-hydrogenation kinetics of aerogels loaded with both MgH₂ and Ticatalyst.
- Obtain pressure-concentration-temperature isotherms to confirm nano-confinement has no effect on the ΔH_{dehvd} of MgH₂.

Anionic Borohydrides

- Exploring variations in reactions conditions to improve yield of lower pressure, full hydrogenation of MgB₂ to Mg(BH₄)₂.
- Continue studies of the catalyzed and un-catalyzed hydrogenation of MgB₂ utilizing MAS ¹¹B NMR to determine the mechanism in order to improve kinetics.

Hydrogenation in Non-Conventonal Solvents

- Maximizing the extended cycling capacity of Ti-doped LiAlH₄ through variation of the dopant concentration and recharging conditions in liquid Me₂O.
- Further detailed evaluation of WTT energy efficiency of the direct synthesis of LiAlH₄ in liquid in collaboration with Argonne National Laboratory.
- Continuing exploration of improving levels of hydrogenation of alane using alternative SCFs and variety of initiators/catalysts.
- Exploring SCF synthesis of Mg(AlH₄)₂.

Patent Applications

1. "Hydrogen Storage Materials, Metal Hydrides, and Complex Hydrides prepared Using Low-Boiling Point Solvents." G. Sean McGrady and Craig M. Jensen U.S. Patent Application 60/945,650, filed 6/08.

2. "Reversible Hydrogen Storage with Magnesium Borohydride." Eva Ronnebro, Godwin Severa, and Craig M. Jensen, U.S. Patent Application 61/093,937, filed 11/08.

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Presentations

1. "Novel Borohydrides for Hydrogen Storage Materials", International Energy Agency, Task 22, Hydrogen Storage Experts Workshop; Rome, Italy, October 6–10, 2009.

2. "High Pressure Hydrogenation of Magnesium Boride to Magnesium Borohydride at Moderate Temperatures", Study of Matter under Extreme Conditions 2009, High Pressure Society of America, Western Caribbean, March 28 – April 3, 2009.

3. "Novel Borohydrides for Hydrogen Storage Materials" International Energy Agency, Task 22, Hydrogen Storage Experts Workshop; JejuIsland, Korea, April 19–23, 2009.

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