First-Principles Modeling of Hydrogen Storage in Metal Hydride Systems

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This presentation does not contain any proprietary or confidential information

Project Overview

- Overall Goals:
 - Provide modeling support for experimental MHCoE group members
 - Predict thermodynamic and materials properties prior to experimental work—guide experimental efforts in new materials development
- Modeling Capabilities:
 - We use *ab initio* and classical methods to compute structural, electronic, thermophysical and chemical properties of materials
 - We can calculate:
 - Enthalpies and free energies of formation
 - Interfacial energies
 - Diffusion pathways and kinetics (transition states)
 - Reaction pathways and kinetics





Project Overview

- Specific Research Areas:
 - Thermodynamic properties of alloys
 - Destabilization alloys such as MgH_2/Si
 - Goal is to compute thermodynamics to aid experimental work and to identify mechanisms to find better materials
 - Calculation of interfacial energies
 - Thin films and nanoparticles potentially have different thermodynamic properties
 - Goal is to compute energetics and thermodynamics to aid in the search for promising nanostructured materials
 - Understanding the role of Ti in catalyzing Na₃AlH₆
 - Mechanisms of Ti in catalyzing hydrogenation and dehydrogenation processes in alanates is still not completely understood
 - Goal is to identify the fundamental mechanisms with the aim of applying them to other, more promising materials
 - Fundamental Processes in Hydrogenation of Al/NaH
 - Hydrogenation of sodium alanate is a model system that is well-studied experimentally

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We aim to apply what we learn about hydrogenation to other more promising materials



Administrative Overview

Timeline

- Project start date: FY05
- Project end date: FY10
- Percent complete: New Start

Budget

- Requested total: \$1.38M
- Cost sharing: 20%
- FY05 \$150k (DOE), \$37.5k (cost share)

Barriers

- Access to adequate computing resources
- Efficient and accurate electronic structure algorithms
- Thermodynamic properties
 from first principles

Partners

- Collaborations with all experimental groups are sought
- Current collaborators:
 - HRL
 - GE
 - U. Hawaii
 - Stanford





Objectives

- Compute thermodynamic properties of metal hydride alloys, e.g., ΔH for:
 - $2MgH_2 + Si \leftrightarrow Mg_2Si + 2H_2$
 - LiBH₄ + 1/2MgH₂ \leftrightarrow LiH + 1/2MgB₂ + 2H₂
 - $4LiH + Si ↔ Li_4Si + 2H_2$
 - Many others
- Compute interfacial properties of hydrides
 - Metal-hydride interfaces, e.g., Al/NaH
 - Surface energies of metal hydrides for nanoscopic hydrides

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 Apply principles learned on model systems, e.g., alanates, to more promising novel materials.



Objectives

- Catalysis processes in hydrides
 - Role of Ti in the decomposition of Na₃AIH₆
 - Inducing reversibility through catalysis of alloy destablized hydrides
- Fundamental processes in hydrogenation AI/NaH and other materials
 - Look for common pathways that might be applicable to other materials
 - What is the role of interfacial transport?





Approach

- First principles density functional theory (DFT) for periodic systems (plane wave)
- Generalized gradient approximation (GGA) for the exchange-correlation functional
- Cluster expansion methods for both ground state and finite-temperature properties
- Transition state finding methods for reaction and diffusion problems
 - Nudged elastic band method
 - Dimer method





Hydride Alloy Structure & Energetics

- We have computed structural (lattice parameters) and energetic (total energies) for a number of different alloys using GGA DFT
- The energies were used to compute enthalpy of reaction for four different destabilization reactions
- Calculations were compared with experiments to assess the accuracy of GGA DFT for these systems
- This is a validation of our technique!





LiBH₄



Calculations are in good agreement with experiments!

Orthorhombic Lattice

| Lattice Parameter | a (Å) | b (Å) | c (Å) |
|----------------------|-------|-------|-------|
| Experimental* | 7.173 | 4.434 | 6.798 |
| DFT-GGA | 7.209 | 4.324 | 6.422 |





* L. Schlapach and A. Züttel, Nature (London) 414, 353 (2001)

MgH_2



Calculations are in good agreement with experiments!

Body centered Tetragonal Lattice

| Lattice Parameter | a (Å) | b (Å) | c (Å) |
|----------------------|-------|-------|-------|
| Experimental* | 4.517 | 4.517 | 3.021 |
| DFT-GGA | 4.466 | 4.466 | 2.992 |



* W.M. Mueller, J.P. Blackledge and G.G. Libowitz, *Metal Hydrides* (Academic Press, New York 1968)



MgB_2



Calculations are in good agreement with experiments!

Hexagonal Lattice

| Lattice Parameter | a (Å) | c (Å) |
|----------------------|-------|-------|
| Experimental* | 3.084 | 3.522 |
| DFT-GGA | 3.068 | 3.520 |





LiH



Calculations are in good agreement with experiments!

NaCl Structure

| Lattice Parameter | a (Å) |
|----------------------|-------|
| Experimental* | 4.083 |
| DFT-GGA | 3.93 |





Mg₂Si



Calculations are in good agreement with experiments!

Cubic Anti-fluorite Structure

| Lattice Parameter | a (Å) |
|----------------------|-------|
| Experimental* | 6.39 |
| DFT-GGA | 6.361 |





Si, Mg, Li







Diamond Structure

| Lattice Parameter | a (Å) |
|----------------------|-------|
| Exp* | 5.43 |
| DFT-GGA | 5.46 |

Hexagonal Lattice

| Lattice Parameter | a (Å) | c (Å) |
|----------------------|-------|-------|
| Exp* | 3.209 | 5.210 |
| DFT-GGA | 3.228 | 5.083 |

BCC Lattice

| Lattice Parameter | a (Å) |
|----------------------|-------|
| Exp* | 3.51 |
| DFT-GGA | 3.401 |

Enthalpy Change for Reactions

$MgH_2(s) + 2LiBH_4(s) \rightarrow 2LiH(s) + MgB_2(s) + H_2(g)$

| | Tabulated* | DFT-GGA |
|--------------------------------|------------|---------|
| ΔH/mol H ₂ (KJ/mol) | 45.96 | 46.2 |

Enthalpies can be calculated with good accuracy from DFT

| | Tabulated* | DFT-GGA |
|------------------------------|------------|---------|
| $\Delta H/mol H_{2}(KJ/mol)$ | 37.24 | 37.1 |

 $2MgH_2(s) + Si \rightarrow Mg_2Si(s) + H_2(g)$

$$LiH(s) \rightarrow Li(s) + H_2(g)$$

| | Tabulated* | DFT-GGA |
|--|------------|---------|
| Δ H/mol H ₂ (KJ/mol) | 181.26 | 183.3 |

$MgH_2(s) \rightarrow Mg(s) + H_2(g)$

| | Tabulated* | DFT-GGA |
|--------------------------------|------------|---------|
| ΔH/mol H ₂ (KJ/mol) | 76.15 | 64.1 |







Conclusions

- DFT is a very effective tool for predicting the enthalpy of reactions
- DFT can be used to compute enthalpies for which no tabulated or experimental data are available, provided the lattice structure is known
- Calculations can be used to screen reactions with high enthalpy without actually conducting any experiments
- We can predict which materials may be suitable as hydrogen storage devices based on calculated enthalpy changes





Reversibility of Destablized Hydrides

- Vajo and coworkers at HRL have found that alloying very stable metal hydrides can substantially increase the release of hydrogen $-2MgH_2 + Si \rightarrow Mg_2Si + 2H_2$
- However, the reverse reaction, while thermodynamically favorable, is not observed
 Mg₂Si + 2H₂ → 2MgH₂ + Si
- We are studying the adsorption and dissociation on Mg₂Si to identify the kinetic barriers





Reversibility of Destablized Hydrides

Initial calculations using plane wave DFT to find adsorption geometries and energetics

 H_2 is on the Si site







Reversibility of Destablized Hydrides

- Four different adsorption sites have been identified
 - Si top
 - Mg top
 - Mg-Si bridge site
 - Hollow site
- All sites lead to a weakly adsorbed (physisorbed) state for H₂









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Conclusions

- Molecular hydrogen adsorbs very weakly through physisorption (van der Waals interactions) on the Mg₂Si surface
- The binding energies are on the order of 0.1 eV
- Dissociation of H₂ on the Mg₂Si surface must have a substantial energy barrier





Future Work

- Calculate dissociation pathways for $\rm H_2$ on $\rm Mg_2Si$
- Calculate binding energies and adsorption barriers for atomic H on Mg₂Si
- Consider catalytic pathways to facilitate the reversibility of $Mg_2Si + 2H_2 \rightarrow 2MgH_2$ + Si



