

## Reversible Hydrogen Storage Materials: Structure, Chemistry, and Electronic Structure

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## Project ID # STP\_36\_Robertson

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# Overview



## **Timeline**

- Project start date
  March 2005
- Project end date
  - February 2010
    - Application to extend project has been made.
- Percent complete 80%

## **Budget**

- Total project funding
  - \$1 600 000 committed
- FY08: \$285K (\$25k supplement)
- FY09: \$100K

## **Barriers**

- Issues addressed
  - A: Understanding chemical response to hydrogen
  - B: Understanding system weight and particle packing
  - C: Theory support to address critical issues blocking experimental progress.

## **Partners**

- Project lead: Sandia National Laboratory & MHCoE
- HRL Laboratory, NIST
- Brookhaven National Laboratory
- Pittsburg and Georgia Tech





- Illinois' main purpose within the MHCoE is
  - Advance the understanding of the microstructural and modeling characteristics of complex hydrides, helping with selection and downselection.
  - Provide feedback and knowledge to partners within MHCoE.
  - Provide more reliable theoretical methods to assess H-storage materials, including key issues affecting materials under study.
  - Help identify downselects, via experimental and theoretical characterization.
  - Help achievement of specific targets and milestones.
  - Provide characterization expertise to other MHCoE and other critical DoE programs, as suggested by Program Officers.

### Some Illinois recent impacts

- Developed 3-D reconstructions and visualization of system and catalyst locale.
- In MgH<sub>2</sub> and MOF systems, complete understanding dispersion of materials.
- Identified new structures that compete in CaB<sub>6</sub>/CaH<sub>2</sub>/Ca(BH<sub>4</sub>)<sub>2</sub> systems, showing amorphous intermediate phases are probable in Ca-based systems (*downselected by SNL*).
- Identified mechanisms for poisoning LiBH<sub>4</sub> systems (*downselected by SNL*).
- Developed quantitative DFT-based theories for complex hydride free energies.
- Developing new surface capabilities to answer fundamental catalysis questions.



80 % complete

70 % complete



- Investigate and model microstructural and chemical changes in hydrogen storage materials
- <u>Task 2: Incorporation and</u> <u>development of new models</u>

 Development of new theoretical models and incorporation into MHCoE work

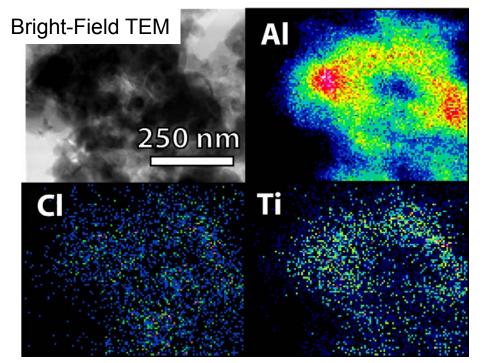
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with Brookhaven National Labs

#### $AICI_3 + 2 \mod \% TiCI_3 + LiAIH_4 \rightarrow AI_{Ti}H_3 + heat -> AI_{Ti} precursor + H_2$

Energy Dispersive Spectroscopy elemental maps of Al<sub>Ti</sub> precursor doped with 2 mol % TiCl<sub>3</sub> before liquid conversion to a reversible material



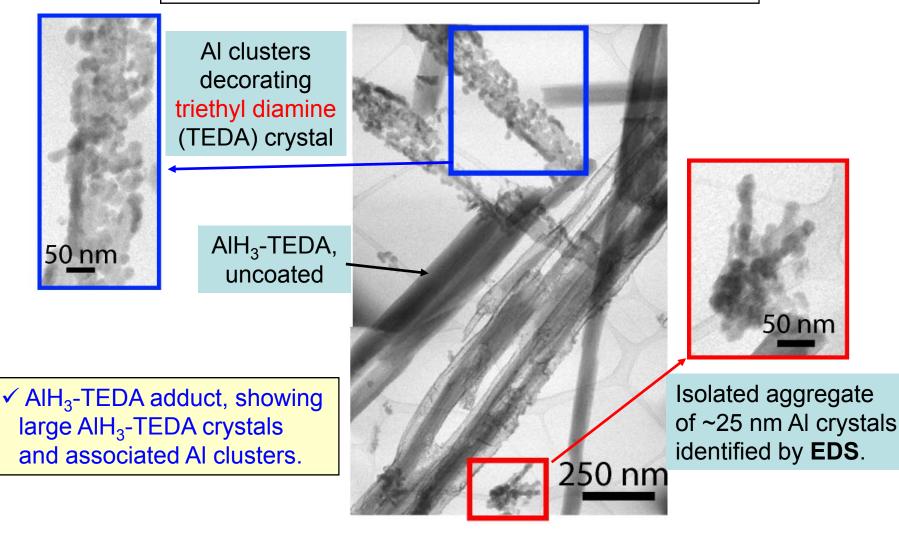
- Extended FY08 analysis to find quantitatively loss of catalyst.
- Quantitative analysis of Ti level (with oxygen excluded) yields 1.2 at% Ti.
- Data indicates a 50% loss of Ti during precursor steps.

✓ In alane precursor, the Ti "catalyst" is well mixed, and perhaps dissolved, before TEDA conversion to reversible material, with 50% Ti catalyst loss.





#### $AI_{Ti}$ precursor + TEDA + $H_2 \leftarrow THF \rightarrow AIH_3$ -TEDA

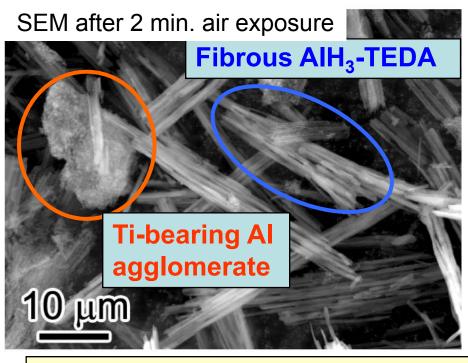




## Morphology and Composition of Hydrided AIH<sub>3</sub>-TEDA with 0.1 at % TiCl<sub>3</sub>



#### $AI_{Ti}$ precursor + TEDA + $H_2 \leftarrow THF \rightarrow AIH_3$ -TEDA



- Fibrous AIH<sub>3</sub>-TEDA

   Mainly C and N detected.
  - -No Ti, placing an upper bound on the Ti level of ~0.25 at% (the estimated detection limit).

#### Al agglomerates

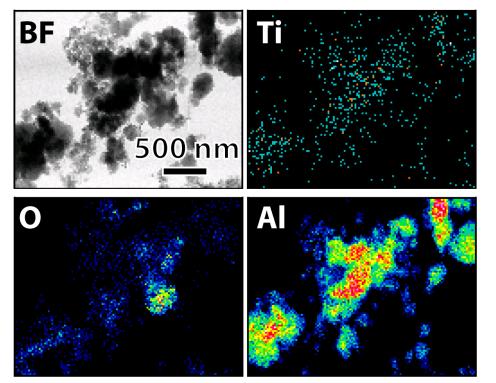
 With 0.1 at% added in total, we observe ~0.4 at% Ti (highly localized, concentrated Ti).

- $\checkmark$  Demonstrates that TEDA scavenges AlH<sub>3</sub> formed on precursor surface.
- ✓ Ti level in the agglomerates increased from the added level of 0.1 at% to 0.4 at%. Consistent with suggested selectivity of AlH<sub>3</sub> by TEDA.
- ✓ Agglomerates suggest TEDA conversion is incomplete in this reaction.



### Morphology and Composition of Dehydrided AIH<sub>3</sub>-TEDA with 2.0 at% TiCl<sub>3</sub>





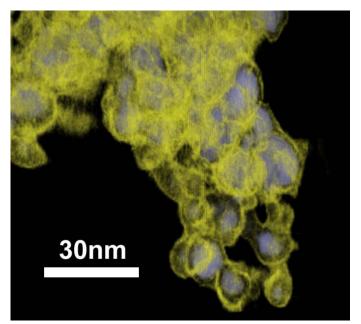
- TEM imaging no air exposure.
- Ti map shows no evidence for local accumulation and redistribution within aggregates of Al crystals.
  - Estimated detection limit 0.2 at% Ti – started at 2.0 at%.
  - Dehydrided material is lacking appreciable Ti.
- Evidence of Ti in precursor-like AI crystals at the *n*m scale is at the detection limit.
- ✓ Even at this level, this constitutes a loss of ~90% of the added Ti, suggesting that very low levels of Ti are needed to catalyze the reaction.



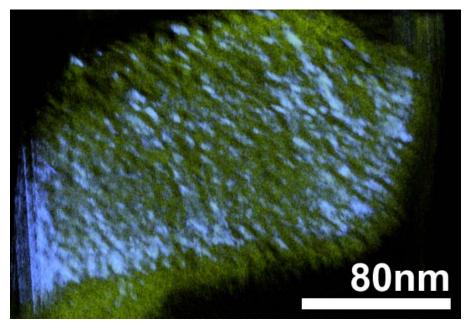


In collaboration with SNL/Berkeley effort

- Electron Tomography has been used to determine Ag particle locations within a metal organic framework (MOF).
- Using Illinois capabilities to support non-MHCoE DoE activities.



Cured Ag-MOF system – Ag particles (blue) – inside framework (yellow).



Uncured Ag-MOF system – Ag particles (including "nanowires" (blue)

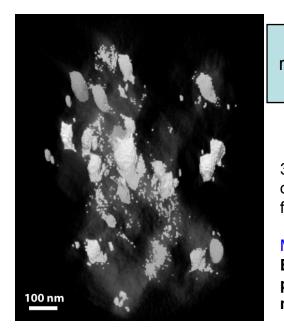
- inside framework (yellow + blue = green).

 ✓ Electron Tomography enables identification of small particle location in MOFs – see movie of three-dimensional reconstructed images.



# Using electron tomography to visualize catalyst dispersal – Ni in MgH<sub>2</sub>/Si

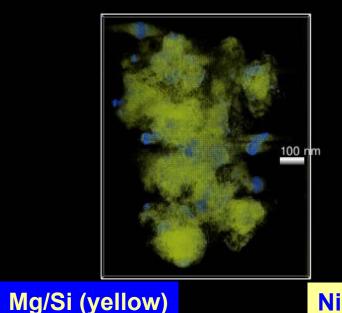




Individual frames used to reconstruct a 3-D image – see movie at poster!

3-D tomographic reconstruction of  $MgH_2$  + Si + 0.05 Ni milled for 5 hrs.

Mg/Si matrix faintly visible. Elongation of spherical particles artifact of limited microscope tilt.



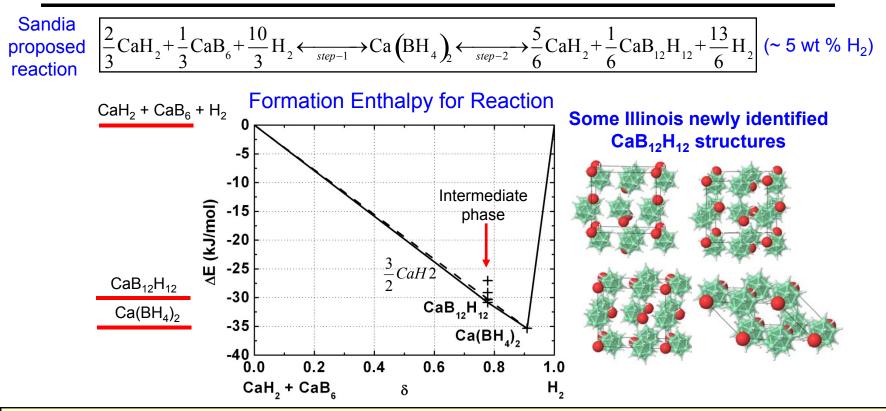
Ni (blue)

- Tomography indicates catalyst dispersal Ni (blue) not well dispersed and remains chemically distinct from the matrix Mg/Si (yellow).
   Raises important question of how it functions as a catalyst.
- ✓ Dissolution of elements by ball milling not guaranteed:
  - Need fundamental understanding of how H interacts with surface and how this is influenced by the introduction of "catalysts".
  - Effectiveness of ball milling to improve kinetics needs to be answered.



Understanding the lack of reversibility of CaB<sub>6</sub>/CaH<sub>2</sub> system: newly predicted competing structures





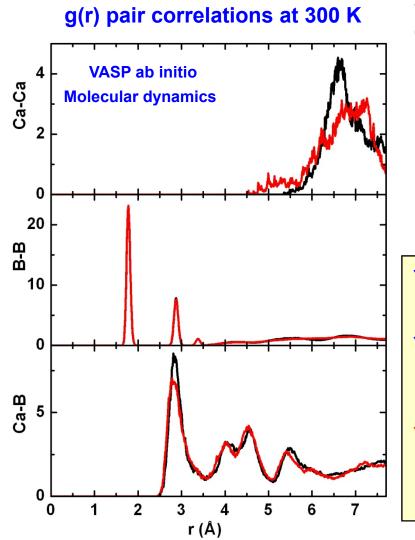
✓ Recent NMR results suggest a  $B_{12}H_{12}$  environment in Li-based system.

- ✓ Illinois DFT results shows CaB<sub>12</sub>H<sub>12</sub> is stable intermediate, but (unlike Li-based systems) with many competing symmetry-distinct phases! Amorphous intermediate phase!
- ✓ Illinois formation energy shows 0.86 (CaH<sub>2</sub> + CaB<sub>6</sub> + H<sub>2</sub>) + 0.14 Ca(BH<sub>4</sub>)<sub>2</sub> = CaB<sub>12</sub>H<sub>12</sub>.
- ✓ Irreversible CaB<sub>6</sub>/CaH<sub>2</sub> system (except at high T and P) but Ca(BH<sub>4</sub>)<sub>2</sub> reversible to amorphous CaB<sub>12</sub>H<sub>12</sub> (possibly mixed phase) matching observation.

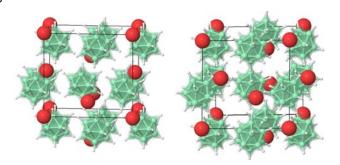


METAL HYDRIDE Structural aspects of CaB<sub>12</sub>H<sub>12</sub> intermediate phases: Illinois newly identified theoretical structures





• Example g(r) for two low-E CaB<sub>12</sub>H<sub>12</sub> structures (cells depicted below) identified theoretically by Illinois, with several others also found.

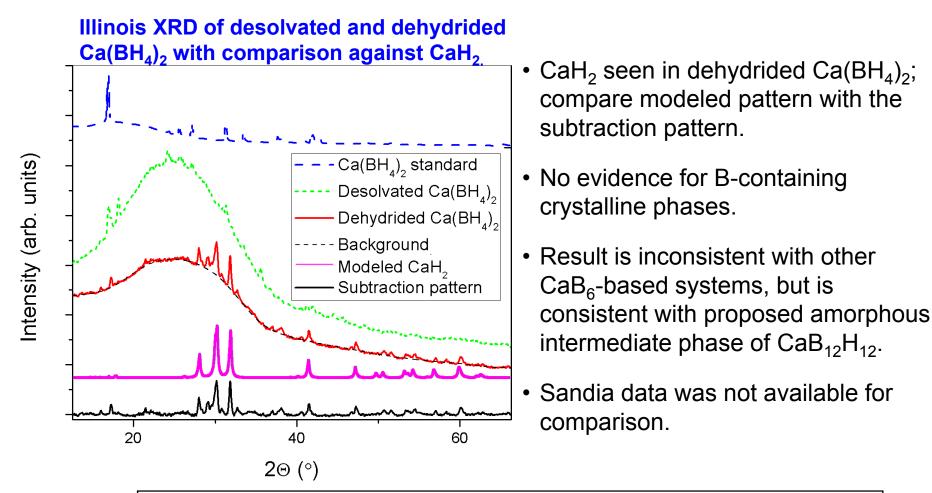


- DFT g(r)'s shows that several symmetrydistinct CaB<sub>12</sub>H<sub>12</sub> phases are (meta)stable!
- Due to strong binding in *closo*-molecule  $(B_{12}H_{12})$ , only Ca-Ca correlation changes with temperature.
- Even at 300 K the Ca-Ca distribution spans a few Angstroms, possibly leading to broad peak in XRD. (At 1000 K, the Ca move from cellto-cell, see Karl Johnson et al. talk.)

Reported in Illinois paper: Wang, Graham et al, submitted







**Proposed reaction :**  $Ca(BH_4)_2 \rightarrow CaB_{12}H_{12} + 5CaH_2 + H_2 (\sim 5 \text{ wt } \% H_2)$ 

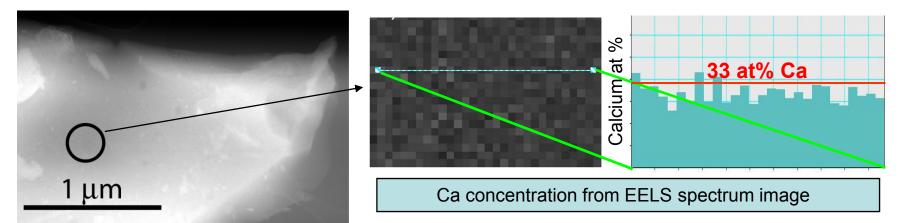
 $\checkmark$  XRD of dehydrided Ca(BH<sub>4</sub>)<sub>2</sub> supports CaH<sub>2</sub> part of proposed reaction.





Dark-field STEM of dehydrided  $Ca(BH_4)_2$  showing the region analyzed with EELS

Proposed reaction: Ca(BH<sub>4</sub>)<sub>2</sub>  $\rightarrow$  CaB<sub>12</sub>H<sub>12</sub> + 5CaH<sub>2</sub> + H<sub>2</sub> (~ 5 wt % H<sub>2</sub>)



Ignoring H (invisible in EELS), expected atomic compositions for proposed phases are:

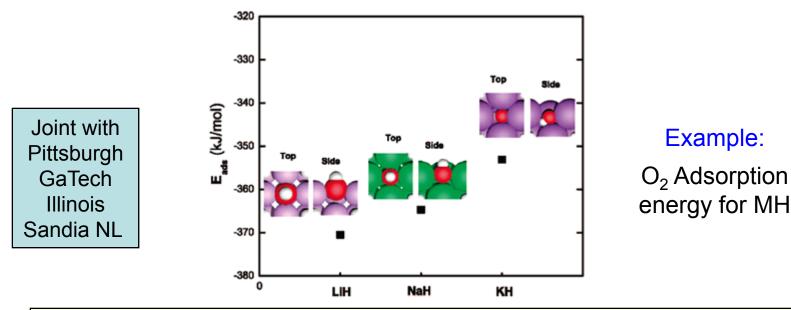
 $Ca(BH_4)_2 = 33 \text{ at\% Ca}, 66 \text{ at\% B} CaB_{12}H_{12} = 8 \text{ at\% Ca}, 92 \text{ at\% B} CaH_2 = 100\% Ca$ 

✓ Composition is the same as parent material Ca(BH<sub>4</sub>)<sub>2</sub>, which means there is no large scale (i.e., > 10 nm) elemental redistribution and no compositional evidence for the intermediate CaB<sub>12</sub>H<sub>12</sub> phase.





Joint theory effort investigated surface reactivities of MH (M= Li, Na, K), products of alkali-metal hexahydride alanates, with H, O, H<sub>2</sub>, O<sub>2</sub>, and H<sub>2</sub>O.

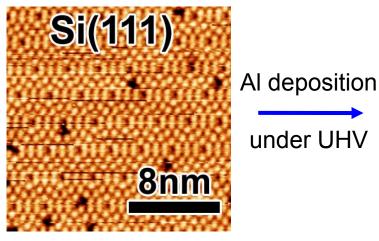


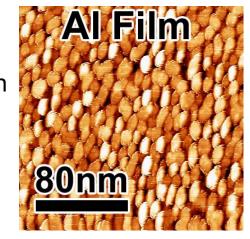
- ✓ DFT results show that LiH has stronger binding to OH, and may contribute to observed differences in rehydriding.
- ✓ But, DFT-based theory also showed that a monolayer of OH forms in less than 10 ms for all alkali metals, which does not account for the observed differences in their ability to form M<sub>3</sub>AlH<sub>6</sub> phase.
- ✓ We find that there is no substantial differences in MH surface reactivities (i.e., poisoning) that correlate with reversibility to M<sub>3</sub>AlH<sub>6</sub> phase.





- Existing work strongly suggests that how hydrogen interacts with pure and contaminated surfaces is crucial to understanding the fundamental processes of hydrogen uptake and release and impact of catalyst species.
  - Thin film of AI grown epitaxially on Si(111) substrate at Illinois.





Si(111) with AI epitaxial layer chosen due to know growth conditions and similarity to other hydrogen-storage materials. (See, e.g., Hasan, et al. Vacuum 41 (1990) 1121.)

✓ Initial work: Made and characterized AI systems for use in examining effects of Ti and surface poisoning on the interaction with hydrogen





#### Sandia NL

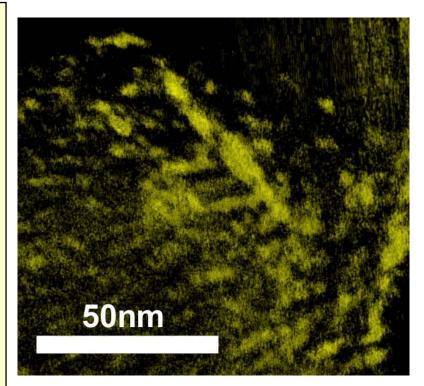
- Answered fundamental questions regarding Ca(BH<sub>4</sub>)<sub>2</sub> systems.
- Provided SNL/Berkley effort with some DFT MgH<sub>2</sub>-cluster/C data.
- Looking at size effects on melting of borohydrides, e.g. in aerogels.
- HRL Laboratories
  - Catalytic particles & tomography.

#### Brookhaven NL

Introductory work on catalyzed AIH<sub>3</sub> systems.

#### • Pittsburgh/GaTech/U. Miss. St. Louis

- From theory, addressed surface reactions on complex hydride reversibility.
- Understanding nature of intermediate M-CaB<sub>12</sub>H<sub>12</sub> phases (M= Mg, Ca, …).



# Close-up of nanowire structure in Ag-MOF – submitted to *Nanoletters*.

*Highlighting Illinois support of DoE partners external to MHCoE efforts.* 





- Analysis of systems using spectrographic, surface, and tomographic capabilities of which we are an integral part
  - Continue work on MOF systems (with SNL)
  - Complete thermodynamic work on  $Ca(BH_4)_2$  but, due to downselect of materials, we will not pursue finding the most effective catalyst.
  - May explore theoretically the effect of catalyst on barriers and uptake.
  - Continue work with BNL on alane systems.
  - Complete AI/Ti surface study (with BNL).
  - Extend collaborations to Physisorption Center.

#### Modeling complex hydride systems –

- With SNL Identify materials not downselected to help improve properties.
- Quantify size and confinement effects on Mg/MgH<sub>2</sub> supported nanoparticles and their adsorption properties (w/ SNL, HRL).
- Apply reaction enthalpy prediction to systems of interest, and use these accurate predictions for van't Hoff plots (w/ GaTech).
- Help understand intermediate phase and reversibility (w/ SNL)
- Complete joint experiment and theory analysis on T-dependent XRD in LiBH<sub>4</sub>. (w/ SNL, UMSL)





- Relevance: We have demonstrated and used a framework for the microstructural analysis of complex hydrides
- Approach: Characterization and theoretical tools are general and can be used with many materials above and beyond complex hydride materials.

#### **Technical Accomplishments and Progress:**

- Demonstrated ability to quantify efficacy of ball-milling and mixing for dispersion of catalyst.
- Addition of powerful tomographic technique.
- Answered questions regarding use of dopants in Ca-based materials, with theory showing limitations of this system and explaining expt.
- Theory provided reliable means to predict quantitatively the reaction enthalpies and structures in various hydrogen-storage materials, including intermediate phases and reversibility (w/ SNL) as planned in 2008-09.
- Surface reactions will be quantified in future.

# **Technology Transfer/Collaborations:** Active interfacing with MHCoE partners, presentations, publications.

 We address critical issues in H-storage materials synthesis and characterization brought to us by MHCoE partners.