

Computational studies of hydrogen interactions with storage materials Chris G. Van de Walle, Lars Ismer, Khang Hoang, Anindya Roy and Anderson Janotti Materials Department, University of California, Santa Barbara

Motivation and challenges

- > Metal hydrides and complex hydrides as hydrogen-storage materials
- > Build understanding of the fundamental mechanisms that control reversible hydrogenation and dehydrogenation processes in hydrides

Systems of interest: Aluminum hydride (AIH₃) and lithium amide (LiNH₂) as reversible hydrogen-storage materials



Objectives

- > Apply state-of-the-art first-principles calculations as well as other simulation techniques to study the microscopic mechanisms that govern mass transport in hydrogen storage materials
- > Determine formation energies and migration barriers of defects related to hydrogen uptake and release
- > Interpret experimental results based on calculated activation energies
- > Investigate the effect of transition-metal additives on hydrogen kinetics
- Study effects of particle size
- > Develop guidelines for designing storage materials with improved storage capacities and favorable kinetics



Aluminum metal phase nucleation in AIH₃

The image shows a cluster of bonded Al atoms in an AIH₃ host crystal. Blue balls: Al; white balls: H.

Methods

- Density functional theory (DFT)
 - generalized-gradient approximation (GGA)
 - screened hybrid functional
 - > J. Heyd, G. E. Scuderia, and M. Ernzerhof,
 - J. Chem. Phys, **118**, 8207 (2003); **124**, 219906 (2006)
- Kinetic Monte Carlo (KMC) model based on first-principles results
 - Enables direct comparison with experiment

Project ID BES015

Dehydrogenation of AIH₃

- > AIH₃ is thermodynamically unstable at ambient conditions, but kinetic effects inhibit dehydrogenation below 100°C
- Contrary to earlier speculation, surface-oxide layer not responsible for metastability
- H.Saitoh *et al.*, Appl. Phys. Lett., **93**, 151918 (2008).
- \succ Decomposition of AIH₃ initiates at localized sites throughout the bulk material, as opposed to nucleation at the surface
 - > S. D. Beattie *et al.*, Chem. Commun. (Cambridge) **2008**, 4448.
- $> V_{\rm H}^+$ and $V_{\rm H}^-$ key defects in the dehydrogenation process
- $> V_{\mu}^{+}$ migrates faster
- Fermi level at the conduction-band minimum ensures the presence of mobile carriers necessary for charge $\hat{\overline{\mathfrak{g}}}$ neutrality



Dehydrogenation kinetics modeled using KMC

- • V_{μ} immobile nucleation centers for AI phase
- • V_{μ}^{+} Random walkers
- Favorable binding energies (trapping energies) as the vacancy clusters grow in size
- > AIH₃ stable for years even without oxide layer KMC results agree with experiment S-shape-like onset of reaction curve does *not* rule out diffusion-limited mechanism
- Simple rules about shapes of

reaction curves can be misleading!



Point defects and diffusion in lithium amide

$LiNH_2 + LiH \leftrightarrow Li_2$	$\mathrm{NH}+\mathrm{H}_2$
-------------------------------------	----------------------------

≻All possible native
defects investigated:
structure, energetics,
migration

>All native defects can be interpreted in terms of H_i^- , V_{H}^- , $(H_2)_i$, Li_i^+ , $V_{\rm li}$, and $V_{\rm NH2}^+$

$\Box_2 \mathbf{N} \mathbf{\Pi} + \mathbf{\Pi}_2$			
Defect	E^f (eV)	E_m (eV)	Complex
H_{i}^{+}	1.28	0.61	
H_{i}^{-}	1.34	0.34	
$V_{\rm H}^-$	0.63	0.71	
$(\bar{\mathrm{H}}_2)_i$	1.75	0.19	
Li_i^+	0.51	0.30	
$V_{ m Li}^-$	0.51	0.20	
$V_{ m Li}^-$ $ m Li_{ m H}^0$	0.48	0.71^{*}	$\mathrm{Li}_{i}^{+} + V_{\mathrm{H}}^{-}$
${ m H}_{ m Li}^0$	1.17	0.61^{*}	$H_{i}^{+}+V_{Li}^{-}$
$V_{\rm NH_2}^+$	0.62	0.87	
$\begin{array}{c}V^+_{\rm NH_2}\\V^0_{\rm NH}\end{array}$	0.40	0.87^{*}	$V_{\rm NH_2}^+ + {\rm H}_i^-$
$V_{ m N}^+$	1.64	0.87^{*}	$V_{\rm NH_2}^+ + ({\rm H_2})_i$
$V_{\rm N}^{-}$	1.77	0.87^{*}	$V_{\mathrm{NH}_2}^+ + 2\mathrm{H}_i^-$



Hydrogen desorption kinetics of lithium amide

- Charge neutrality maintained by Li-related Frenkel pairs
- > Two mechanisms for dehydrogenation of $LiNH_2$ suggested:

Mechanism I (bulk): self-diffusion of (H_i^+, V_H^-) rate-limiting process



Decomposition mechanisms of LiBH₄ and LiAIH₄

- Investigated effects of transition-metal additives



Future directions

- MgH_2 , Mg_2 (Fe/Ni)-hydride, LiAlH₄, etc.
- \succ Role of transition-metal doping in MgH₂:
- the dilute limit

Overall summary and impact

Consideration of charged point defects and impurities (and not just neutral) states) in hydrides: an essential physics issue previously not addressed > State-of-the-art first principles calculations to accurately describe defects > Multiscale modeling to address realistic system sizes and temperature > Broader conclusions: shapes of reaction curves, particle-size dependence

References:



Mechanism II (surface): self-diffusion of V_{μ}^{-} rate-limiting process



> Dominant mechanism for decomposition depends on particle size (specific surface area): explains variation of activation energy with duration of ball milling > Hydrogen Frenkel pairs at the core of decomposition mechanism

> Comprehensive study of point defects and dehydrogenation mechanisms

Role of hydrogen-related Frenkel pairs in dehydrogenation kinetics > Systematic studies in a range of materials: LiBH₄, Li₄BN₃H₁₀, LiNH₂,

Formation energies and migration barriers in Mg₂-TM and MgH₂ matrix in

L. Ismer, A. Janotti, and C. G. Van de Walle, Appl. Phys. Lett. 97, 201902 (2010). L. Ismer, A. Janotti, and C. G. Van de Walle, J. Alloys Compd. 509S, S658 (2011). K. Hoang, A. Janotti, and C. G. Van de Walle, Angew. Chem. Int. Ed. 123, 10352 (2012). K. Hoang, A. Janotti, and C. G. Van de Walle, Phys. Rev. B 85, 064115 (2011). K. Hoang and C. G. Van de Walle, Int. J. Hydrogen Energy **37**, 5825 (2012). K. Hoang, A. Janotti, and C. G. Van de Walle, Phys. Chem. Chem. Phys. 14, 2840 (2012).