# **Advanced Hydroxide Conducting Membranes**

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**Project ID: FC123** 



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# Overview

### Timeline

- Project Start: August 1 2014
   Project End: July 31 2015
- Percent Complete: 70

### **Budget**

- Total Project Budget (K): 500
- Funding Received in FY 15 (K): 500
- Total DOE Funds Spent (K): 340

No cost share 68% spent as of April 10, 2015

### **Barriers**

- B. Cost
- C. Electrode performance
- A. Durability

### **Project lead**

Los Alamos Nat. Lab.

(Project Management, Ionomer Synthesis, MEA Fabrication and Fuel Cell Testing)

Yu Seung Kim (PI)

Kwan-Soo Lee



### **Collaborators/Interactions**

#### **Subcontractors**

- Sandia Nat. Lab. (Membrane Synthesis)
- Lawrence Berkeley Nat. Lab. (Property & Modeling)

#### Collaborators/Interactions

- Rensselaer Polytech. Institute (Membrane Synthesis)
- Solvay (Material Supply)
- Ion Power (Material Supply)
- IRD Fuel Cell
   (Fuel Cell Testing)
- National Institute of Advanced Industrial Science and Technology (Stability Modeling)

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# Relevance

### **Objectives**

Develop (1) *chemically stable hydroxide-conducting anion exchange membranes* and (2) *solvent processable perfluorinated ionomers* and (3) *modeling approaches (HOR, ORR and polarization behaviors)* to demonstrate high performance/durable alkaline membrane fuel cells.

### **Technical Barriers**

- Low hydroxide conductivity
- Poor cationic functional group & polymer backbone stability
- Mechanical-instability to make thin films
- Low quality perfluorinated ionomer dispersions

### **Targets and Current Status**

Characteristics	Units	2014 status <sup>a</sup>	Target <sup>b</sup>	Current status <sup>c</sup>
Areal resistance	$\Omega \cdot cm^2$	0.17	≤ 0.10	0.06 - 0.08
IEC loss after >100 h 0.5 M NaOH at 80 $^\circ$	%	10	No decrease	No decrease
Peak power density of AEMFC	W/cm <sup>2</sup>	<b>580</b> (14 mg <sub>Pt</sub> /cm <sup>2</sup> )	> <b>600</b> (0.4 mg <sub>Pt</sub> /cm <sup>2</sup> )	<b>330</b> (0.4 mg <sub>Pt</sub> /cm <sup>2</sup> )

<sup>a</sup> Y.S. Kim, Resonance-Stabilized Anion Exchange Polymer Electrolytes, US DOE 2014 Annual Merit Review, June, 16-20, 2014 <sup>b</sup> The interim project target was set for general fuel cell applications <sup>c</sup> as of April 10, 2015



## **Approach:** Anion exchange membranes

### Polymer Backbone

 Developing polymers without ether or electron-withdrawing groups in the polymer backbone

### Aryl-ether cleavage reaction<sup>a</sup>



### Our approach

- ✓ Ether-free poly(phenylene) (SNL)
- Ether-free polystyrene-bpoly(ethylene-co-butylene)-bpolystyrene (RPI)

### Cationic Functional Group

 Replacing benzyl trimethyl ammonium (BTMA) with more alkaline stable cations

### S<sub>N</sub>2 degradation of BTMA<sup>b</sup>



## Our approach

- ✓ Hexamethyl ammonium (SNL, RPI)
- Phenyl guanidinium (LANL, SNL)







### FY14 LANL Perfluorinated Ionomer\*





\* D.S. Kim et al. *Macromolecules*, 46, 19, 7826-7833 (2013)

## **Approach:** Performance modeling



D T



#### **Boundary conditions:**

- (1) 1 4 no flux for  $OH^-$ ,  $CO_3^{-2}$  and  $HCO_3^-$  except for  $H_2O$  at boundaries 1 and 2.
- At the boundary 1 (aCL/GDL interface), the water content is fixed at  $\lambda = 10$ (2) At the boundary 2 (cCL/GDL interface),  $\lambda = 10$  (RH = 100%) or  $\lambda = 4$  (RH 40%)

#### **Properties:**

- (1) Thickness of aCL and cCL =  $10 \mu m$  and that of AEM =  $25 \mu m$ .
- (2) Cell potential = 0.8 V and T =  $80 \degree \text{C}$
- (3)  $[CO_2]$  in the aCL and cCL are the same (0.005% to 50% ambient  $[CO_2]$ )  $[CO_2]$  in the AEM is set to be zero (assuming AEM has low  $CO_2$  tolerance). Note: ambient  $CO_2$  pressure = 0.0004 atm (ambient  $[CO_2] = 1.6*10^{-5}$  mol/L).
- (4)  $OH^{-}$  conductivity = 0.08 S/cm for 100% hydrated AEM.

#### **Electrode kinetics:**

$$\begin{array}{ll} @ \text{ anode } & i_a = a_{P_l} i_{0,a} \left(\frac{C_{H_2}}{C_{H_2}^{ref}}\right)^{0.5} y_{OH^-} \exp\left(\frac{\alpha_a \eta_a F}{RT}\right) & \eta_a = -\Phi_e - V_a^{eq} & V_a^{eq} = \frac{RI}{nF} \ln(y_{OH}^2) \\ @ \text{ cathode } & i_c = a_{P_l} i_{0,c} \left(\frac{C_{O_2}}{C_{O_2}^{ref}}\right) \lambda \exp\left(\frac{-\alpha_c \eta_c F}{RT}\right) & \eta_c = \Phi_{cell} - \Phi_e - V_c^{eq} & V_c^{eq} = 1.23 + \frac{RT}{nF} \ln(y_{OH}^2) \\ & a_{P_l} = 10^5 \text{ cm}^{-1} & i_{0,HOR} = i_{0,ORR} = 3 \text{ mA/cm}^2 & C_{H_2} = C_{O_2} = 10 \text{ mol/m}^3 & C_{H_2}^{ref} = 16 \text{ mol/m}^3 \\ \bullet \text{ LATIONAL LAROPATIONS} \end{array}$$

## **Accomplishments:** Synthesized rigid polymers



# Accomplishments: Synthesized flexible polymers

#### **Nafion®-FA-TMG Series**



- Perfluorinated (PF) control AEM (LANL & SNL)
  - **Progress** Polymer synthesis: 100% Characterizations: 100%

- Guanidinium functionalized stable PF anion exchange polymer
- Strength
   Low water uptake
   Excellent hydrophobicity ideal for ionomeric binder

#### Weakness Processibility for ionomer dispersion Instability of amide group



- SEBS based AEM (RPI)
- Progress
   Polymer synthesis: 100%
   Characterizations: 80%
- Highly conductive and stable polymers prepared from C-H borylation and coupling reactions
  - **Strength** High conductivity Good polymer stability Mechanical toughness
  - Weakness Processibility for ionomer dispersion Relatively high water uptake

#### **PF Alkyl Amide Series**



- PF polymers having alkyl amide group (LANL)
- **Progress** Polymer synthesis: 80% Characterizations: 0%
- PF polymers prepared from new chemistry
- Strength (expected) Better conductivity Good amide stability Improved processibility for ionomer dispersion
- Weakness







For AEM and dispersion appearance: see back-up slides 21, 22

## Accomplishments: Membrane conductivity/resistance

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#### Conductivity Summary

Membrane code	IECª	OH <sup>-</sup> cond. <sup>b</sup>	Areal Resistance <sup>c</sup> (film thickness)	
	meq./g	mS/cm	$\Omega \cdot  ext{cm}^2$ ( $\mu$ m)	
ATM-PP	1.7	40	0.17 (60)	
MRH	2.0 - 2.4	40 - 60	0.06-0.08 (30)	
AR	1.9 – 2.1	41 – 54	0.10-0.15 (22)	
SEBS-QA	1.0 – 1.9	29 – 45	0.06-0.07 (35)	

<sup>a</sup> by titration; <sup>b</sup> measured at 30°C; <sup>c</sup> measured at 80°C

- Reduced areal resistance of FY 15 AEMs compared with that of ATM-PP control.
- Less resistance variation with current density for MRH and SEBS-QA AEMs, indicating better water management
- Highlight: Achieved FY15 areal resistance milestone (< 0.1 Ω cm<sup>2</sup>) for MRH and SEBS-QA AEMs











## Accomplishments: Membrane mechanical properties

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Summary of Mechanical Properties

Membrane code	Water uptake <sup>a</sup>	Stress <sup>b</sup>	Strain <sup>b</sup>
	(wt. %)	(MPa)	(%)
ATM-PP	83	28	20
MRH	126	28	23
AR	120	36	50
SEBS-QA	220	4	>300

 $^{\rm a}$  measured at 30°C;  $^{\rm b}$  Hydroxide form AEMs measured at 50°C, 50% RH

- Improved mechanical properties of AR series AEMs compared with ATM-PP control
- More than 300% strain for SEBS-QA AEMs at 50°C, 50% RH; > 20% strain at 0% RH (Back-up slide 23)
- Durable MEAs could be fabricated from all FY15 AEMs

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## Accomplishments: Membrane stability



- Improved alkaline stability of FY15 AEMs compared with ATM-PP control from ex-situ test.
- Confirmed alkaline stability of SEBS-QA AEMs under AMFC operating conditions
- Highlight: Achieved FY 15 stability milestone for AR and SEBS-QA: < 10% loss after 500 h, 0.5M NaOH at 80°C AEMs (May 30, 2015)
- Next step: Complete In-situ and ex-situ life test for MRH, SPG, and AR series AEMs







## Accomplishments: Amide group stability of PF polymers



## Accomplishments: On-going fuel cell performance test



- Significant AMFC performance depending on ionomer structure; Best performance using MRH ionomers among other hydrocarbon ionomers
- Next step: Fuel cell performance optimization using PF ionomers

# Accomplishments: CO<sub>2</sub> effect modeling



Current density @ 0.8 V versus  $[CO_2]$ 

(OH<sup>-</sup> & CO<sub>3</sub><sup>-2</sup>) mole fraction versus [CO<sub>2</sub>]



 Even the small amount of 50% of atmospheric CO<sub>2</sub> concentration (~ 0.02% CO<sub>2</sub>) can significantly reduce the current density and deplete the OH<sup>-</sup> species by CO<sub>3</sub><sup>-2</sup> in the MEA.

Modeling results on CO<sub>2</sub> effect



Experimental results on CO<sub>2</sub> effect



# **Accomplishments:** Potential & water distribution modeling

Electrolyte phase potential distribution along the **MEA** thickness



AFM



OH<sup>-</sup> transport is governed mainly by migration due to potential gradient. However, there is less variation in the ionic potential across MEA with higher  $[CO_2]$ , reducing OH<sup>-</sup> flux.





aCL

 $O_2 + 2H_2O + 4e^- \rightarrow 4OH^ 2H_2 + 4OH^- \rightarrow 4H_2O + 4e^-$ 

cCL





# **Collaborations/Interactions**

Institutions	Туре	Extent	Role and Importance
Los Alamos National Laboratory	Federal Laboratory	Major	<ul> <li>DOE Hydrogen and Fuel Cell Program (Prime)</li> <li>Synthesis and characterization of AEMs and ionomer dispersion</li> <li>Membrane, electrode and fuel cell performance test</li> </ul>
Sandia National Laboratories	Federal Laboratory	Major	<ul> <li>DOE Hydrogen and Fuel Cell Program (Sub)</li> <li>Synthesis and characterization of poly(phenylene) AEMs</li> <li>Access to the state-of-the-art poly(phenylene) AEMs</li> </ul>
Lawrence Berkeley National Laboratory	Federal Laboratory	Major	<ul> <li>DOE Hydrogen and Fuel Cell Program (Sub)</li> <li>Characterization and modeling of AEMs</li> </ul>
Rensselaer Polytechnic Institute	Academia	Major	<ul> <li>RPI Internal Funding (Outside DOE program)</li> <li>Synthesis and characterization of SEBS and poly(biphenylene) AEMs</li> </ul>
Solvay/Ion Power	Industry	Medium	<ul> <li>Supply of non-standard PFSA precursors</li> </ul>
IRD	Industry	Minor	<ul> <li>DOE Incubator Program</li> <li>50 cm<sup>2</sup> MEA testing</li> </ul>
National Institute of Advanced Industrial Sci. and Technol.	Foreign National Laboratory	Medium	<ul> <li>LANL Cooperative Research and Development Agreements</li> <li>DFT modeling</li> </ul>



# **Remaining Challenges and Barriers**

#### Membrane

 Alkaline stability: Long term fuel cell performance loss associated with AEMs and possible mitigation strategy should be investigated.

#### Ionomeric binder

 Quality of ionomer dispersion: Ionomer particle size in liquid dispersion plays major role in interfacial reaction of electro-catalysts. Current ionomer dispersion does not provide best three-phase interface structure in the electrode layer.

#### Membrane electrode assembly

 Interfacial compatibility: Interfacial compatibility/adhesion between hydrocarbon membrane and PF ionomer-bonded electrode needs to be improved for high AMFC performance.

#### AMFC performance

- (Bi) carbonate contamination: Performance loss due to (bi) carbonate contamination may be problematic for end-use applications. Mitigation strategy needs to be developed.
- Catalyst HOR activity: Cation adsorption to anode catalysts significantly lowers the overall performance. → leveraged research efforts with DOE Incubator project.
- Non-precious metal catalysts: Extensive works with MEA fabrication and fuel cell testing are required. → leveraged research efforts with DOE Incubator project.
- Water management: Flooding and water management issues may still remain.



# **Proposed Future Work**

Task	Description	Expected completion day
1	Property optimization of AR series AEMs (RPI)	June 30, 2015
2	Synthesis and characterization of sulfone (or ketone) phenyl guanidinium functionalized poly(phenylene) AEM (SNL, LBNL and LANL)	June 30, 2015
3	Development of high quality PF alkyl amide polymer dispersion (LANL)	June 30, 2015
4	AMFC test under O <sub>2</sub> and air conditions (initial and long- term test) (LANL, IRD)	July 31, 2015
5	Incorporate the GDL and flow channels into the existing AMFC model (LBNL)	July 31, 2015
6	Manuscript submission and Patent filing (RPI, SNL, LBNL, LANL)	October 30, 2015



# Summary

- FY15 MRH, AR and SEBS-QA series AEMs showed excellent conductivity, mechanical properties and alkaline stability compared with ATM-PP control.
  - Areal resistance < 0.1 Ω cm<sup>2</sup> was achieved for MRH and SEBS-QA series AEMs
  - Less than 5% degradation under > 1M KOH at 80°C was achieved for MRH, AR and SEBS-QA series
- FY15 PF ionomer with alkyl amide linkage was designed to increase amide stability under basic conditions; Fuel cell performance measurement using these ionomers is on-going.
- Developed a membrane+electrode model that takes into account the effect of CO<sub>2</sub> contamination on the AMFC performance; Combining with experimental data, benefits of higher temperature operation in the presence of CO<sub>2</sub> was demonstrated.



# **Technical Back-Up Slides**



## **Solution Cast AEMs**



#### SEBS-QA Series (30 - 50 µm thick)



Obtained thin, tough and uniform thickness membranes from solution cast techniques.



## **Ionomer Dispersions**

### Nafion<sup>®</sup>-FA-TMG

#### (2.5 wt.% dispersion)



### Nafion<sup>®</sup>-FA-TMG

(2.5 wt.% dispersion)



### ATM-PP

(2.5 wt.% dispersion)



**PF Alkyl Amide** 

(1 wt.% dispersion)



#### **Fine dispersion** Particle size < 0.1 µm

Rensselaer

Sandia

National laboratories

MRH (1 wt.% dispersion)



AR (5 wt.% dispersion)



- 1 to 2.5 wt.% anion exchange ionomer dispersions were prepared.
- High quality dispersion using patent pending technology\*

\*Y.S. Kim, LANL Idea 15-00037

## **Effect of RH on Mechanical Properties of SEBS-QA AEMs**

Cell temperature: 50°C; RH: 0, 50, 90%; Number of samples: 2 Equilibrium time: 0% and 50% RH  $\rightarrow$  40 min; 90% RH  $\rightarrow$  60 min; Load: 0.5 MPa/min; testing time: 2 to 5 hours / sample



 Like other ion exchange membranes, tensile strength increased and elongation at break of SEBS-QA AEMs decreased as RH increased.



## Governing equations for ion and water fluxes



 $CO_3^{-2}$  fraction  $HCO_3^{-1}$  fraction OH<sup>-</sup> fraction Electro-neutrality  $y_{OH^{-}} = \frac{C_{OH^{-}}}{C}$   $y_{CO_{3}^{-2}} = \frac{C_{CO_{3}^{-2}}}{C}$   $y_{HCO_{3}^{-}} = 1 - y_{OH^{-}} - 2y_{CO_{3}^{-2}}$  $C_{OH^{-}} + 2C_{CO_{3}^{-2}} + C_{HCO_{3}^{-}} = C_{ammonium^{+}} = C_{t}$  $A \equiv OH^{-}$  $B \equiv CO_3^{-2}$ Migration Diffusion Streaming-current  $C \equiv HCO_3^-$ **OH** flux  $N_A = -D_A C_t \nabla y_A + \frac{F D_A C_t}{PT} y_A \nabla \Phi - \frac{D_A C_t}{PT} y_A \xi_A \nabla \mu_w$  $w \equiv water$  $D_i$ : diffusion coefficient  $CO_3^{-2}$  flux  $N_B = -D_B C_t \nabla y_B + \frac{2FD_B C_t}{PT} y_B \nabla \Phi - \frac{D_B C_t}{PT} y_B \xi_B \nabla \mu_w$  $\xi_i$ : electro-osmotic drag coefficient  $HCO_{3}^{-} \text{ flux } N_{c} = -D_{c}C_{t}\nabla(1-y_{A}-2y_{B}) + \frac{FD_{c}C_{t}}{PT}(1-y_{A}-2y_{B})\nabla\Phi - \frac{D_{c}C_{t}}{PT}(1-y_{A}-2y_{B})\xi_{c}\nabla\mu_{w}$ aCL AEMcCL Electro-osmotic drag **Back diffusion** back diffusion water flux  $N_w = (\sum_{i} \xi_i N_i) - D_w C_t \nabla \lambda = \xi_A N_A + \xi_B N_B + \xi_C N_C - D_w C_t \nabla \lambda$ electroosmosis  $O_2 + 2H_2O + 4e^- \rightarrow 4OH^ 2H_2 + 4OH^- \rightarrow 4H_2O + 4e^-$ 24