

# HyMARC: A Consortium for Advancing Solid-State Hydrogen Storage Materials

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Livermore, CA  
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*Enabling twice the energy density for onboard H<sub>2</sub> storage*



**Project ID: ST127**

This presentation does not contain any proprietary, confidential, or otherwise restricted information

# Overview

## Timeline

**Project Start Date: 9/17/2015**

**Phase 1 end date: 9/30/2018**

## Barriers

- A. System Weight and Volume**
- E. Charging/Discharging Rates**
- O. Lack of Understanding of Hydrogen Physisorption and Chemisorption**

## Budget

- FY15 DOE Funding: \$1,250K**
- FY16 DOE Funding: \$2,430K**
- FY17 DOE Funding: \$2,823K**
- FY18 DOE Funding: \$3,025K**
- Total DOE Funds Received: \$9,528K**

## Partners

- Sandia National Laboratories**
- Lawrence Livermore National Laboratory**
- Lawrence Berkeley National Laboratory**



# Relevance/Objective: accelerate discovery of breakthrough storage materials by providing capabilities and foundational understanding

**Foundational understanding** of phenomena governing thermodynamics and kinetics limiting the development of solid-state hydrogen storage materials

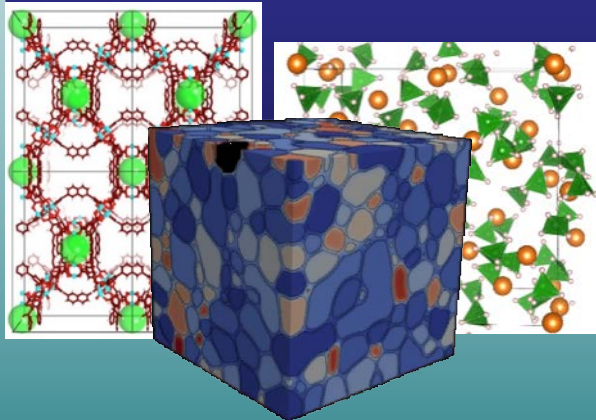
HyMARC will deliver **community tools and capabilities:**

- **Computational models and databases** for high-throughput materials screening
- **New characterization tools and methods** (surface, bulk, soft X-ray, synchrotron)
- **Tailorable synthetic platforms** for probing nanoscale phenomena

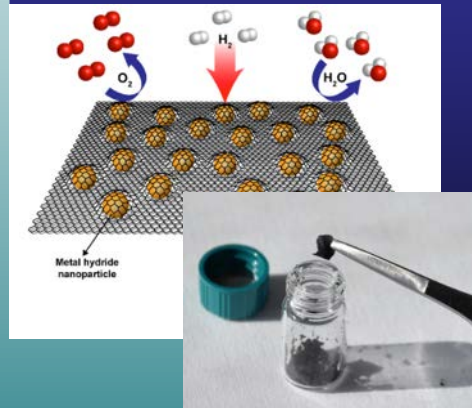
## Core Lab Team



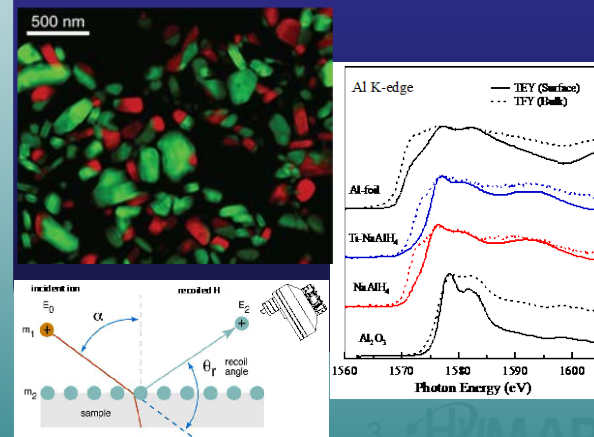
## Theory, simulation, & data



## Controlled synthesis



## In situ characterization



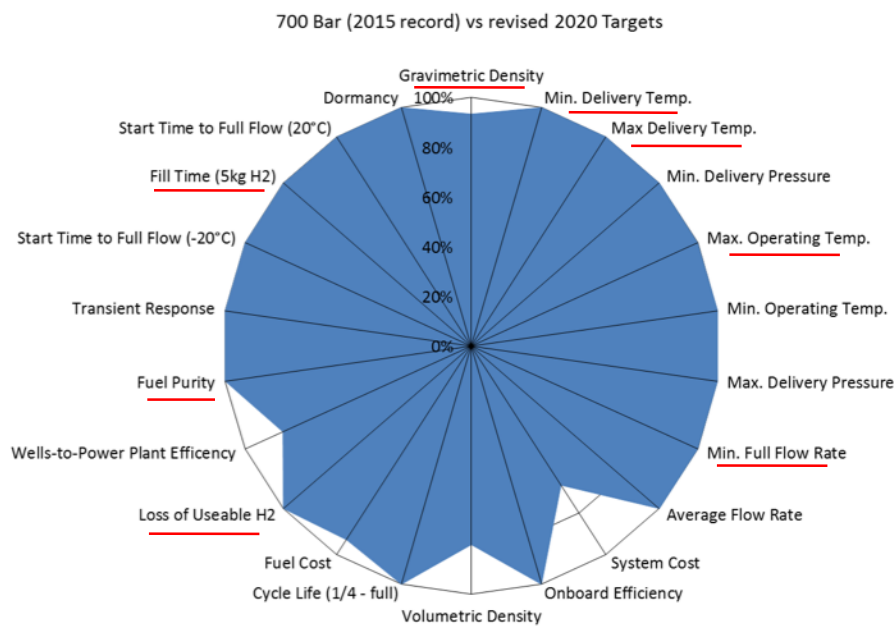


# Relevance: HyMARC is accelerating materials discovery and moving the bar relative to compressed H<sub>2</sub>

## Accelerating materials discovery

- **Strategy assessments:** identified most promising material improvement strategies
- **Missing/inaccurate data:** e.g. thermodynamic data essential for material assessment
- **Modeling tools:** filling major gaps in understanding of key processes
- **Enabling Seedling Projects by providing:**
  - Access to experimental resources essential to their success (e.g., hi-P reactors and PCT)
  - Computational modeling in support of experiments (outside Seedling budget)
  - Assisting with data interpretation (e.g., computational spectroscopy)

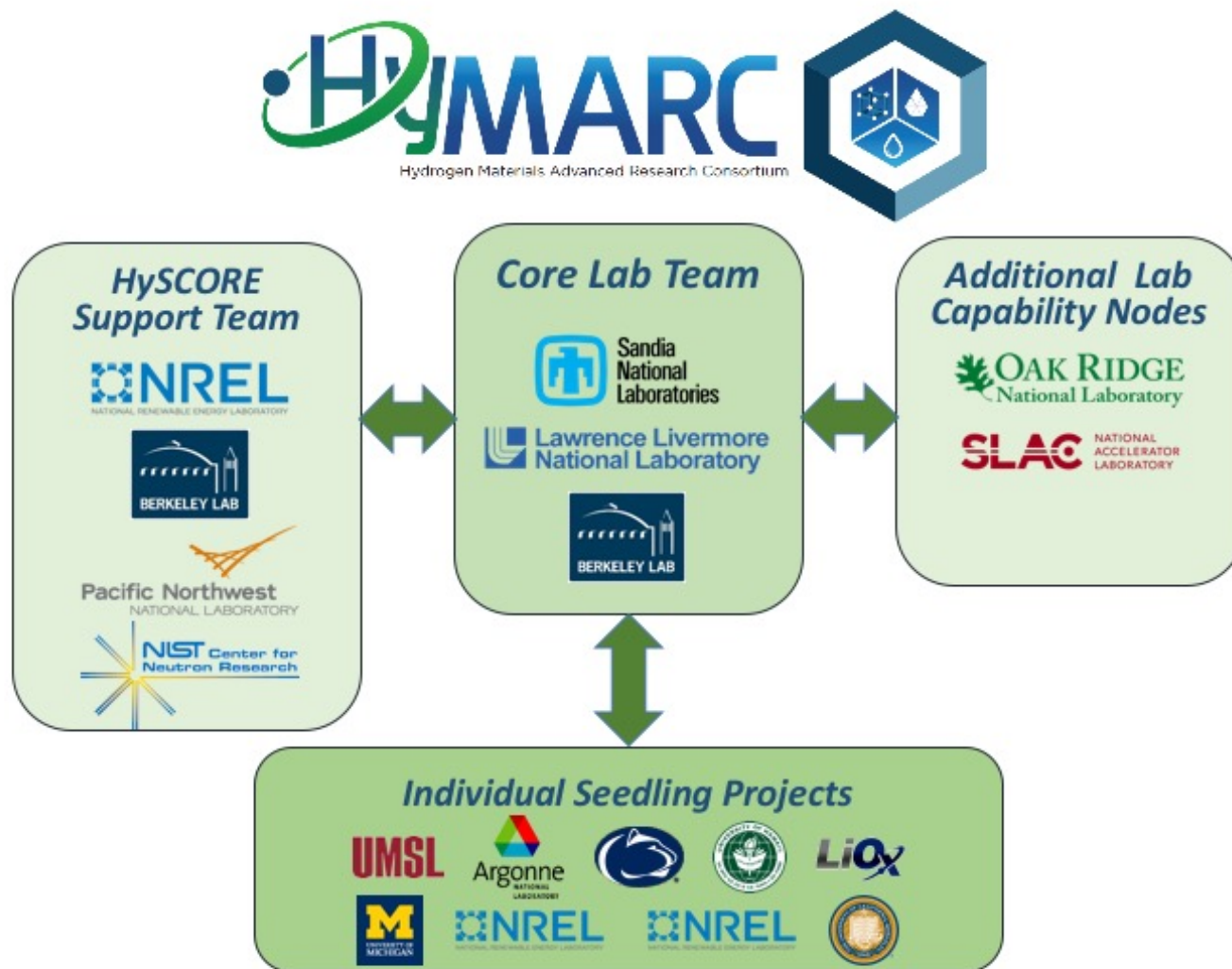
*Material development strategies we evaluated indicate progress toward several DOE targets*



## Moving the bar for specific materials or strategies

- Interface engineering: Li<sub>3</sub>N@(6nm-C) H<sub>2</sub> cycling T reduced by >180 °C (bulk is 430 °C)
- Nanoconfinement (porous host): Mg(BH<sub>4</sub>)<sub>2</sub>@(6-nm C) H<sub>2</sub> desorption T reduced > 100 °C
- Nanoencapsulation: Mg(BH<sub>4</sub>)<sub>2</sub>@rGO >10 wt% (record for nanoscale hydride)
- Thermodynamics tuning: U. Hawaii Seedling project (MgB<sub>2</sub> etherates): advances to Phase 2

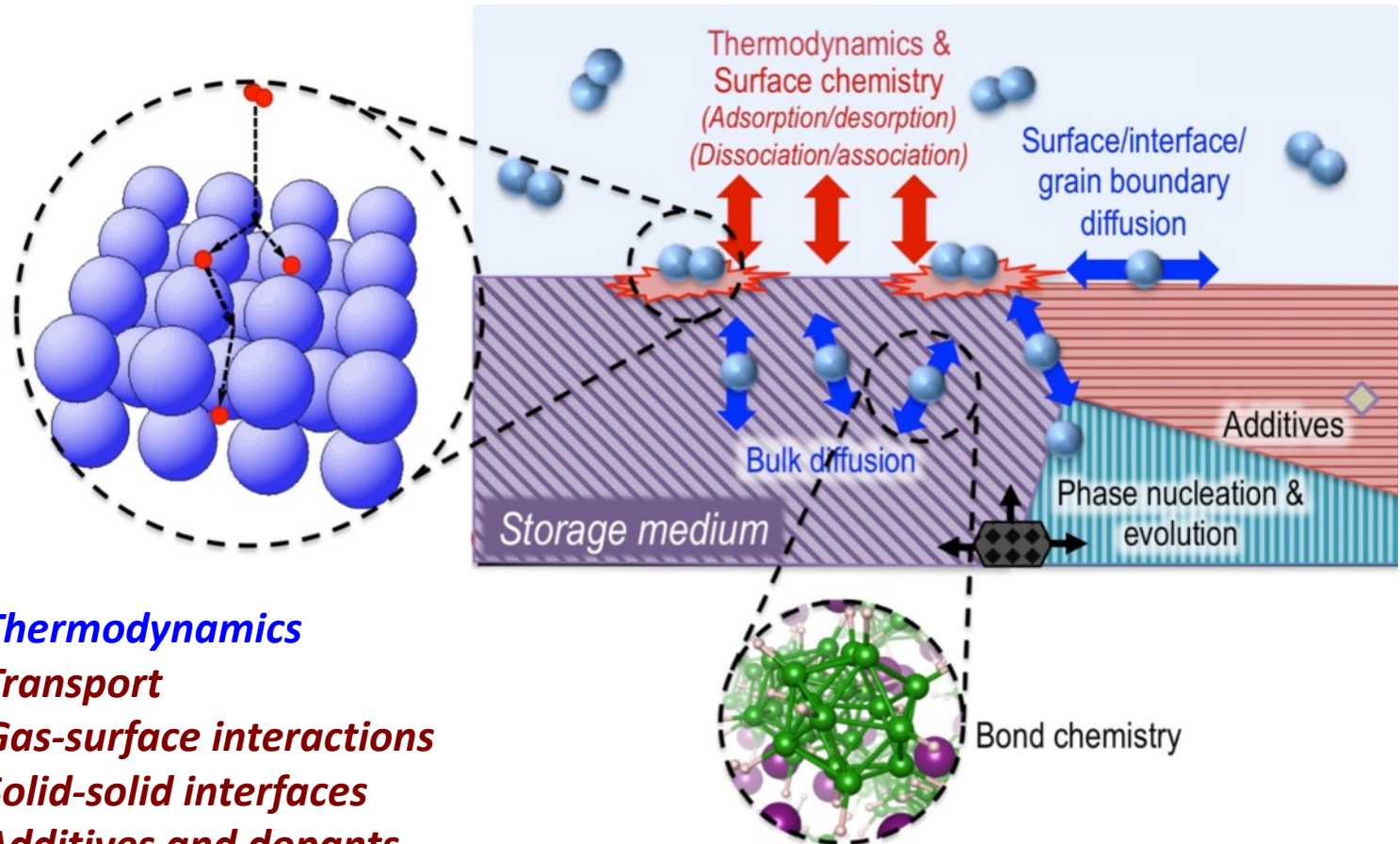
# Approach: The HyMARC Core Lab team addresses foundational research objectives and supports Seedling projects



*Enabling twice the energy density for hydrogen storage*

# Approach: HyMARC/Core Team tasks target 1) thermodynamics and 2) all phenomena potentially influencing reaction kinetics

Effective thermal energy for H<sub>2</sub> release:  $\Delta E(T) = \Delta H^\circ (T) + E_a$   
Thermodynamics Kinetics



**Task 1: Thermodynamics**

**Task 2: Transport**

**Task 3: Gas-surface interactions**

**Task 4: Solid-solid interfaces**

**Task 5: Additives and dopants**

**Task 6: Materials informatics**

# Accomplishments overview: technical themes highlighted in the consortium partner presentations

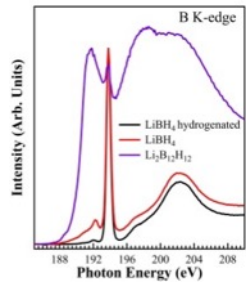
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- **New capabilities** developed or enhanced:
  1. Modeling
  2. Characterization
  3. Synthesis
- **BES User Facilities and international collaborations** build foundational understanding
- **Seedling Project success** enabled and enhanced
- **Communication:** making HyMARC results accessible to the hydrogen storage community
  - Workshops
  - Sorbent Perspective
  - Nanohydride review

# Accomplishment: modeling tools that now cover all relevant length scales and many important phenomena

Atomic/molecular  
(0 – 1 nm)

Computational  
Spectroscopy

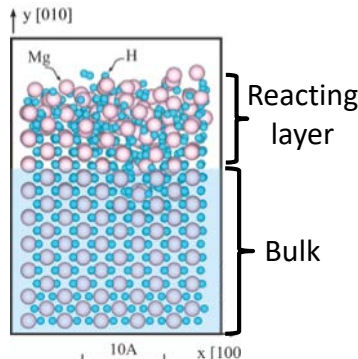
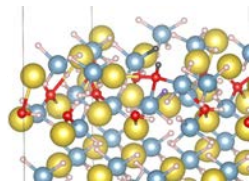


Example:  
NaAlH<sub>4</sub> surface  
chemistry: role of oxide



Molecular/micro  
(0.5 – 2 nm)

Surface chemistry  
Interatomic potentials

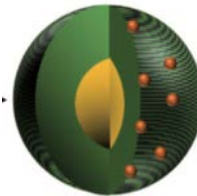
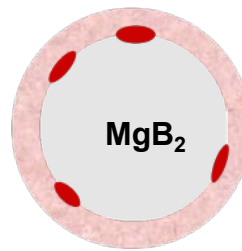


Example:  
Time-dependent  
simulations of  
MgH<sub>2</sub> formation



Mesoscale  
(2 - 100 nm)

Nucleation kinetics  
Phase microstructures

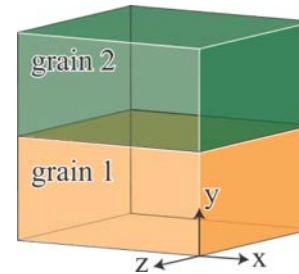


Example:  
Nano-alloying of  
Ni-doped Mg



Grains  
(≤ 10 μm)

Grain boundaries  
Particle size effects  
Stress/strain

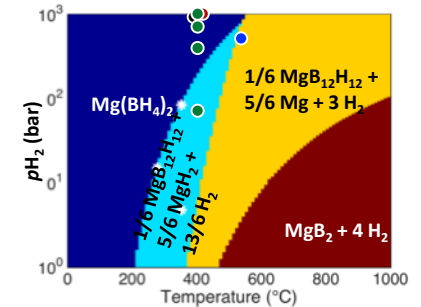


Examples:  
H diffusion in PdH<sub>x</sub>  
Diffusion in NaBH<sub>4</sub>



Macroscale/Bulk

Thermodynamics



Example:  
Mg(BH<sub>4</sub>)<sub>2</sub> phase diagram



10<sup>-10</sup>

10<sup>-8</sup>

10<sup>-6</sup>

10<sup>-4</sup>

10<sup>-2</sup>

Length (m)



# Accomplishment: characterization tools expanded and extended to in-situ, in-operando probing and mesoscale resolution

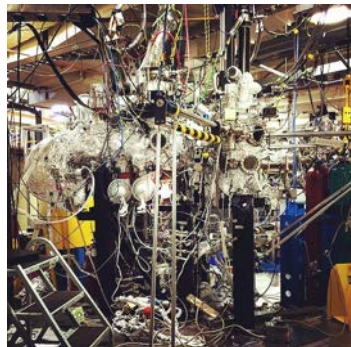
Atomic/molecular  
(0 – 1 nm)

Molecular/micro  
(0.5 – 2 nm)

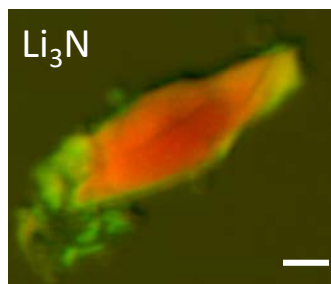
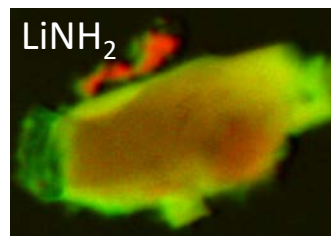
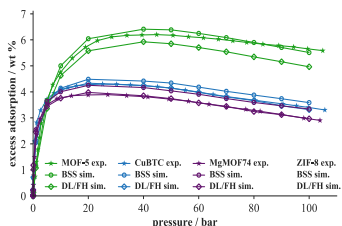
Mesoscale  
(2 - 100 nm)

Grains  
( $\leq 10 \mu\text{m}$ )

Macroscale/Bulk



Microporosimetry/BET



Ultrahigh Pressure Reactor  
(1000 bar)



H-D exchange

AP-XPS  
ALS/BL 11.0.2

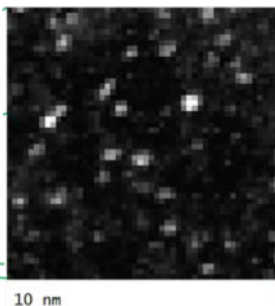


XAS In-situ flow cell  
(1 bar, max. 250°C)



Lab-based  
AP-XPS

He bubbles seen by  
AC-TEM  
STEM res. 63 pm



STXM (30 nm res.)  
LBNL/ALS



$10^{-10}$

$10^{-8}$

$10^{-6}$

$10^{-4}$

$10^{-2}$

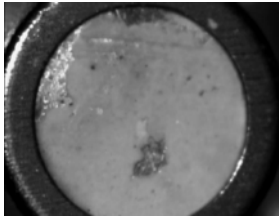
Length (m)

# Accomplishment: >400 hours of synchrotron and neutron beam time at 5 international user facilities in the US, Canada, and Japan

	Shifts (8 h)	Samples
<b>ALS Berkeley, CA (BES user facility): <span style="color: red;">Approved Program continues</span></b>		
STXM	9	Partially dehydrog. Li <sub>3</sub> N and LiNH <sub>2</sub> +2LiH (N K edge)
XAS	6	TiF <sub>3</sub> /TiCl <sub>3</sub> doped MgB <sub>2</sub> ; oxidation of MgB <sub>2</sub> ; B-doped C aerogels
<b>ALS Berkeley, CA (BES user facility): <span style="color: red;">General User Proposal approved</span></b>		
AP-XPS	12.5	Mg(BH <sub>4</sub> ) <sub>2</sub> <i>in situ</i> desorption: pure & partially oxidized samples
<b>SLAC/SSRL Stanford, CA (BES user facility): <span style="color: red;">General User Proposal approved</span></b>		
TXM	3	MgB <sub>2</sub> and Mg(BH <sub>4</sub> ) <sub>2</sub>
<b>UVSOR Myodaiji, Okazai, Japan (<span style="color: red;">all travel expenses covered by UVSOR</span>)</b>		
STXM	6	partially hydrogenated MgB <sub>2</sub> and partially dehydrogenated Mg(BH <sub>4</sub> ) <sub>2</sub> (B K edge)
CLS/REIXS	14	MgB <sub>2</sub> ; TiCl <sub>3</sub> /TiF <sub>3</sub> -doped MgB <sub>2</sub> ; nano-MgB <sub>2</sub> ; B-doped C aerogels
<b>NIST Gaithersburg, MD</b>		
NVS, INS		Mg(BH <sub>4</sub> ) <sub>2</sub> , MOF-74 samples
Diffraction		M-closoborates (e.g B <sub>12</sub> H <sub>12</sub> ) samples
<b>ORNL/Spallation Neutron Source/VISION: <span style="color: red;">General User Proposal approved</span></b>		

# Accomplishment: New sample formats for encapsulated complex hydrides, MgB<sub>2</sub> nanoparticles, graphene nanostructures

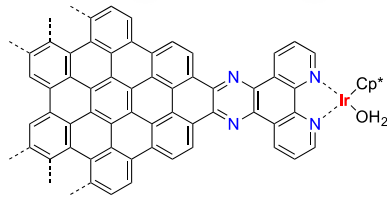
Atomic/Molecular  
(0 – 1 nm)



Mg(BH<sub>4</sub>)<sub>2</sub> film  
on Au for LEIS  
measurements



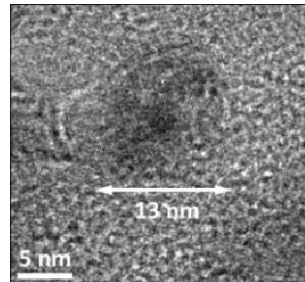
Molecular and  
microscales  
(0.5 – 2 nm)



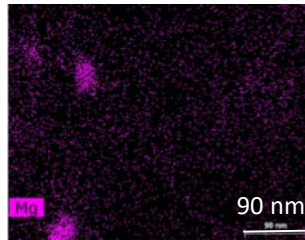
Model systems:  
GNR+(H<sub>2</sub> dissociation  
catalyst)



Mesoscale  
(2 - 100 nm)



Mg(BH<sub>4</sub>)<sub>2</sub>@rGO

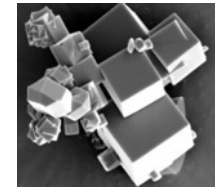


MgB<sub>2</sub> nanoparticles  
Encapsulation  
Strain effects  
Nanoscaling

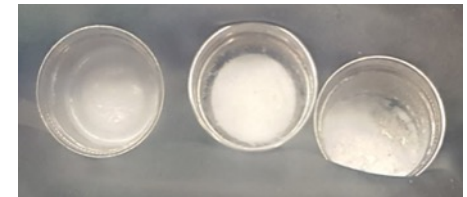


Grains  
(up to ~ 10 μm)

Macroscale/Bulk



High-purity MOFs for  
model validation



New thermodynamics:  
Liquid-phase Mg(BH<sub>4</sub>)<sub>2</sub>



10<sup>-10</sup>

10<sup>-8</sup>

10<sup>-6</sup>

10<sup>-4</sup>

10<sup>-2</sup>

Length (m)

# Accomplishment: Two major publications document Go/No-go milestones

## HyMARC FY17/Q2 Go/No-go Milestone

*Rank improvement strategies for sorbents. Decision criterion: select 2 with greatest potential for increasing  $\Delta H^\circ$ . Top strategies:*

- *Open metal sites in MOFs*
- *Lewis acid/Lewis-base sites*

## HyMARC FY18/Q4 Go/No-go Milestone

*Rank improvement strategies for hydrides. Decision criterion: select 2 with greatest potential for reducing effective  $\Delta H$*

(article addresses a major strategy considered in the Go/No-go)

Submitted to *Energy & Environ. Sci.*

**“An Assessment of Strategies for the Development of Solid-State Adsorbents for Vehicular Hydrogen Storage”**

Topics include:

- Usable gravimetric and volumetric capacities
- The importance of binding strength
- Theoretical calculations of H<sub>2</sub> physisorption
- Considerations for adsorbent synthesis and characterization
- Revisiting the results of the 2010 HSCoE final report
- Perspectives on current material strategies

Submitted to *Chem. Rev.*

**Nanostructured Metal Hydrides for Hydrogen Storage**

Topics include:

- Classes of nanostructured metal hydrides
- Synthesis routes
- Structure
- Morphology
- Mechanistic effects



# Collaborations and Coordination: HyMARC is actively collaborating with Seedling Projects and facilitating their research

## *Improving communications: Designated point-of-contact identified for each Seedling Project*

- **Development of Magnesium Boride Etherates as Hydrogen Storage Materials** (U. Hawaii)



- Instability in  $MgB_2$  B sheets explained (LLNL modeling investigation)
- High-P hydrogenation, XRD, and FTIR performed for 43  $MgB_2$ (etherate) samples

- **“Graphene-wrapped” Complex Hydride Materials** (Argonne National Lab.)



- Go/No-Go 3 samples processed at Sandia (March 2018)

- **Surface-Functionalized Mesoporous Carbons for Thermodynamic Stabilization and Reversibility of Metal Hydrides** (Univ. Missouri, St. Louis)



- XPS, FTIR, publication (Chem. Mater.); samples shipped to us

- **Developing A Novel Hydrogen Sponge with Ideal Binding Energy and High Surface Area for Practical Hydrogen Storage** (Penn State Univ.)



- Measured  $H_2$  isotherms (low- and high-P)

- **Electrolyte Assisted Hydrogen Storage Reactions** (Liox Power)



- 1) High-P experiments and sample characterization; 2) Go/No-go tests underway

- **ALD Synthesis of Novel Nanostructured Metal Borohydrides** (NREL)



- $Mg(BH_4)_2$  nanoparticle samples sent to NREL for ALD coating

- **Fluorinated COFs: A Novel Pathway to Enhance Hydrogen Sorption** (NREL)



- Modeling study; paper submitted

- **Optimized Hydrogen Adsorbents via Machine Learning & Crystal Engineering** (U. MI)



- Discussions on crystal engineering of OMS in MOFs

- **Super-Metalated Frameworks as Hydrogen Sponges** (UC Berkeley)



- Discussions concerning use of high-pressure reactor and PCT

# Collaborations with external research groups are vigorous

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## DOE BES User Facilities

- **Dr. A.J. (Timmy) Ramirez-Cuesta (SNS/ORNL):** neutron vibrational spectroscopy
- **Dr. Simon Bare (SLAC/SSRL):** synchrotron measurements

*Advice and assistance writing/submitting General User Proposals for these facilities*

## Academia and Government

- **Prof. Torben Jensen (Aarhus University, Denmark):** metal borohydride intermediates
- **Prof. Stefan Kaskel (Technical University Dresden):** high surface-area MOFs
- **Prof. H.-C. Yu (U. Michigan):** Phase-field model development
- **Prof. Nobuhiro Kosugi, Director, UVSOR:** synchrotron measurements
- **Prof. Eun Seon Cho, KAIST:** experimental measurements of strain effects

*Several manuscripts published, accepted, or underway*

# HyMARC is making the results of its research available to the community using multiple communication channels

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## Conferences and workshops

- ACS Spring 2018 symposium “Hydrogen Energy: Production, Storage & Application,” March 13 – 17, 2018, New Orleans, LA (V. Stavila, T. Autrey, organizers)
- 2017 Advanced Light Source Users meeting “Surmounting Fundamental Challenges to Vehicular Hydrogen Storage,” Oct. 4 – 5, 2017, Berkeley, CA (M. Allendorf, F. El Gabaly, J. Guo organizers)
- Machine learning workshop: “Application of Machine Learning and Data Analytics for Energy Materials Network Consortia 2018,” May 2 – 3, 2018, Livermore, CA (B. Wood, organizer)

## Presentations

- 14 contributed presentations at conferences (ECS, APS, MRS, ACS, GRC)
- 20 invited presentations

## Journal publications

- 8 published since last AMR
- 2 invited reviews submitted (*Energy Environ. Sci.* and *Chem. Rev.*)

# Progress toward Milestones

Project Milestone	Type	Task Completion Date				Progress Notes
		Original Planned	Revised Planned	Actual	% Complete	
<p>Use QMC, DFT, and force fields to compute H<sub>2</sub> binding and select appropriate levels of theory for MOFs.</p> <p><b>Revised milestone: Compute H<sub>2</sub> binding curves with different computational methods for model MOFs to establish protocol for accurate physisorption calculations</b></p>	PM	12/31/17	<b>3/31/18</b>		70%	LLNL and LBNL are working on finite-size/extended system corrections based on MOF-74. Study of different DFT functionals on model noncovalent systems published in recent sorbent review.
Sensitivity analysis of local binding and second-sphere effects	PM	3/31/17		9/30/17	100%	Completed by Maciek Haranczyk (LBNL) for open metal sites in MOFs. Results to appear in sorbent perspective
Rank improvement strategies for sorbents. Decision criterion: select 2 with greatest potential for increasing $\Delta H^\circ$	GNG	3/31/17		3/31/17	100%	Strategy and results reported at AMR. Manuscript for submission to <i>En. Env. Sci.</i> in progress
Modify LEIS instrument to enable laser-induced thermal desorption	PM	6/30/17		6/30/17	100%	Demonstrated feasibility of ion beam-based desorption technique
Evaluate additive/composite strategies for improving effective $\Delta E$	PM	9/30/17	<b>9/30/18</b>		75%	Evaluation of multiple strategies in progress
Prototype hydride surface and interface chemistry kinetic models	M	9/30/17		9/30/17	100%	
Amorphous phases and defects model formalism	PM	12/31/17		12/31/17	100%	



# Progress toward Milestones (cont.)

Project Milestone	Type	Task Completion Date				Progress Notes
		Original Planned	Revised Planned	Actual	% Complete	
Sensitivity analysis of morphology and microstructure	PM	3/31/18		3/31/18	100%	Completed analysis, reported in LLNL AMR slides
Rank improvement strategies for hydrides. Decision criterion: select 2 with greatest potential for reducing effective $\Delta H$	GNG	3/31/18		3/31/18	50%	Nanoscaling and Dopants addressed experimentally
Rank improvement strategies for hydrides. Decision criterion: select 2 with greatest potential for reducing effective $\Delta H$	GNG	3/31/18		3/31/18	50%	Nanoscaling and Dopants addressed experimentally
Parameterize integrated kinetic model for representative B-N-Al-hydrides	PM	6/30/18			75%	Thermodynamic parameters completed. Some kinetic parameters have been calculated for Na-Al-H and Mg-B-H
Compute sorbent isotherms from QMC data using CoRE database of MOFs	PM	9/30/18			0%	Not started
Milestone delayed until Phase 2 due to lack of personnel with requisite expertise						
Public release of codes, databases, synthetic protocols, characterization methodologies optimized for storage materials	M	9/30/18			0%	Not started. Release of codes is a time-consuming process that will be difficult to complete for codes that as of the date of this report are not ready.
Revised language: Public release of databases, synthetic protocols, characterization methodologies optimized for storage materials						

# Remaining Challenges and Barriers

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**Access to ALS facilities remains limited, particularly for beam lines that can probe light elements such as boron and for in-operando measurements**

*An ALS General User Proposal was accepted to allow access to the AP-XPS. We initiated a relationship with SLAC/SSRL (AP-XPS and other tools) and with UVSOR (STXM for B) to provide alternatives*

**A single database framework that can integrate experimental and theory data, as well as past data archived within the DOE Hydrogen Storage Materials Database**

*A Data Management Plan will be developed in the remainder of FY18 with assistance Energy Materials Network that will employ common database formats and activities. This will make the best use of limited resources and facilitate data sharing.*

**The broad diversity of HyMARC activities and potential storage materials are challenging to address**

*We are planning a new organizational structure for HyMARC Phase 2 that will employ “Focus Groups” that address hot topics or major roadblocks with small, dynamic teams, with the intention of making rapid progress by concentrating resources in specific areas*

# Proposed future work

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- **To the extent possible, integrate submodels to enable more comprehensive approach to predicting hydride kinetics and behavior**
- **Initiate Data Management Plan with assistance of Kristin Munch (NREL) and using platform and utilities previously developed for other EMNs**
- **Initiate renewal process for ALS Approved Program (2<sup>nd</sup> 3-years)**
- **Complete and submit as many publications as possible to fully document the Phase 1 progress**
- **Submit additional General User Proposals to SLAC/SSRL and ORNL/SNS to expand access**
- **Update HyMARC web site to include HySCORE capabilities and updated publication list**
- **Initiate model integration activity, in collaboration with AMPE code developers at LLNL**
- **Write and submit proposal for HyMARC Phase 2**
- **Write and submit final report**

***Any proposed future work is subject to change based on funding levels***

# Summary

## **Model development is moving at a fast pace; several new capabilities now operational**

- Finite-T hydride thermodynamics
- Grain boundary model
- Particle morphology model
- Solid mechanics/strain effects model
- Surface chemistry model

## **Advanced characterization capabilities and methods are now on line**

- H/D exchange, AC-TEM, and AP-XPS capabilities at Sandia now available to HyMARC
- A vigorous experimental program at BES and other user facilities is providing many new insights
- Surface chemistry instrumentation suite probes all relevant length scales
- New user proposals approved for access to ORNL/VISION, SSRL

## **New synthetic methods provide materials and data targeted at specific phenomena**

- rGO demonstrated as hosts for complex hydrides
- Effects of high-pressure cycling on sorbents evaluated
- New insights into metal borohydride chemistry

## **These new capabilities are generating foundational understanding that is changing our view**

- Role of surface oxide and  $-OH$
- New strategies for altering hydride thermodynamics (e.g., strain, nanoencapsulation)
- Effects of high-P cycling on MOF sorbents
- Understanding of which bond-breaking events must be targeted for additive development

***Robust collaborations within HyMARC, with Seedling projects, and international partners are ensuring an interdisciplinary approach focused on cutting-edge materials discovery***



**We are grateful for the financial support of EERE/FCTO and for technical and programmatic guidance from Dr. Ned Stetson, Jesse Adams, and Zeric Hulvey**



*Enabling **twice the energy density** for onboard H<sub>2</sub> storage*



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# Technical Back-Up Slides

# Background for FY17/Q2 Go/No-go Milestone (3/31/2017)

**Rank improvement strategies for sorbents. Decision criterion: select 2 with greatest potential for increasing  $\Delta H^\circ$ . Literature reports and modeling also taken into account.**

## Downselected strategies

1. **Open metal sites (OMS):** It is well established for MOFs that  $H_2$  binds more strongly to metal cations with incomplete coordination spheres than to other structural features where only weak physisorption occurs. For example, the highest  $Q_{st}$  reported for a MOF without an OMS was 9.5 kJ/mol as of 2012, whereas it is 15.1 kJ/mol for a MOF with OMS and there are several in the 10 – 12 kJ/mol range (see M. P. Suh et al. *Chem. Rev.* 2012, 112, 782–835).
2. **Lewis-acid/base sites:** OMS in MOFs are Lewis acid sites and are considered as a separate strategy. A related strategy is to incorporate electron-deficient atoms such as boron into porous carbon or other porous materials. This strategy appears promising;  $Q_{st}$  values up to ~9 kJ/mol are observed for porous graphene oxide (G. Srinivas et al. *J. Mater. Chem.*, 2011, 21, 11323) and theory predicts that B-doping may raise the adsorption energy of  $H_2$  from 4–8 kJ/mol for pure carbon materials to 15–35 kJ/mol; see e.g. Y. Xia et al. *J. Mater. Chem. A*, 2013, 1, 9365). Experimentally,  $Q_{st}$  of 12.47 kJ/mol has been reported for a 7.2% B-doped microporous carbon (see Chung et al. *JACS* 2008, 130, 6668). This strategy has not been thoroughly investigated, however.

**Other strategies:** these were not highly ranked, due either to their small documented effect on  $Q_{st}$  or lack of evidence supporting their use for viable storage materials.

1. **Frustrated Lewis pairs:** Incorporating both Lewis acid and Lewis base atoms to create frustrated Lewis pairs to polarize  $H_2$  has not been thoroughly explored.
2. **Polarization by functionalized MOF linkers:** ab initio calculations predict that adding electron-donating groups (e.g.  $CH_3$  or  $NH_2$ ) to the aromatic rings of MOF linkers increases  $Q_{st}$  by only ~15%. Electron-withdrawing groups tend to decrease  $Q_{st}$ , although systematic studies of  $N_2$  uptake with IRMOF-1-X (X=halide) indicate slightly higher  $Q_{st}$  for X=I than X=F (S. T. Meek et al. *J. Phys. Chem. C* 2012, 116, 19765). Larger  $Q_{st}$  enhancements were observed by adding amide groups to the linker rings, but the largest increase was 0.7 kJ/mol (13%; see Z. Wang et al. *Chem. Eur. J.* 2010, 16, 212).
3. **Brønsted acid sites:** There are examples of metal-exchanged zeolites with adsorption enthalpies as high as 17.5 kJ/mol and there are neutron data suggesting values as high as 20-40 kJ/mol might be achievable. However, zeolites are not viable storage materials and Brønsted acid sites in MOFs are rare and are less accessible than in zeolites.
4. **Phase-change materials:** this is a little-investigated strategy for MOFs and one for which it is difficult to predict how much  $Q_{st}$  would be increased, since one cannot readily compare with an analogous, but structurally rigid, structure. There is evidence that structures with large breathing modes bind  $H_2$  more tightly in the “closed” form, but only a few examples exist. This strategy may have more value for increasing the amount of useable hydrogen, as recently demonstrated by Long et al. for methane storage (Mason et al. *Nature* 2015, 527, 357).